



Reuse of recovered recyclate from infusible thermoplastic carbon fiber composites to reduce virgin material usage in reprocessed composites

Radwa Youssef^{1,*}, Monika von Monkiewitsch², Gerhard Zeigmann¹, Ioannis Manolakis³, Peter Wierach^{1,2}, Steffen Opitz⁴  and Souvik Chakraborty⁴ 

Abstract

This work presents a novel combinatory approach in relation to our previous work, exploiting the previously demonstrated recovery of Elium matrix from CFRP laminates (simulating CFRP waste) and its incorporation as Elium monomer replacement for subsequent CFRP composite manufacture with reduced raw monomer material usage. Our previous efforts provided a proof of concept for the addition of neat pre-polymerised Elium recyclate from polymer production waste, whereas it was acetone-extracted CFRP Elium matrix used here as monomer substitute. The recovery was achieved by dissolving Elium CFRP in acetone to separate the fibers and recover the matrix, followed by incorporating the recovered matrix in granulate form (hereafter recovered recyclate) in the resin mixture for manufacturing new composites. This was carried out by substituting a fraction of the Elium monomer with recovered recyclate at different weight fractions (2.5 - 7.5 wt%) and dissolving it in the Elium monomer before polymerization. It was possible to integrate up to 5 wt% of recovered recyclate during composite manufacturing, resulting in laminates with comparable matrix molecular weight and thermal properties, improved thermal stability and comparable mechanical performance to the virgin counterpart.

Keywords

CFRP, recycling, fiber/matrix interface, reusing recyclates

Introduction

Global market of carbon fiber reinforced thermoplastic composite has been growing steadily, and the demand is expected to grow, which is anticipated to increase at the compounded annual growth rate (CAGR) by 7.8% during the period from 2022 to 2030.¹ The exceptional characteristics of these CF based thermoplastics composites which combines high strength with being lightweight make them suitable for a wide range of applications, especially within the automotive and aerospace sectors.² The surge in growth comes with concerns regarding the waste generated at the end-of-life. Thus, the development and continuous improvement of recycling techniques is crucial to product sustainability. The processing cost along with the carbon footprint can be reduced through the implementation of an effective recycling process, retaining recovered components' (fiber and resin) mechanical as well as chemical properties.

Mechanical, thermal, and chemical recycling are generally the main recycling categories for CFRP that are

currently available. These recycling techniques generally result in a compromise of either the mechanical properties of the recovered fibers or the chemical structure of the matrix.

¹Institute for Polymer Materials and Plastics Engineering, Clausthal University of Technology, Clausthal-Zellerfeld, Germany

²Department of Multifunctional Materials, Institute of Lightweight Systems, German Aerospace Center (DLR), Braunschweig, Germany

³Department of Life Sciences, Atlantic Technological University, Co Sligo, Ireland

⁴Department of Sustainability Technologies, Institute of Lightweight Systems, German Aerospace Center (DLR), Braunschweig, Germany

*Presently at: Department of R&D, Etimex Primary Packaging GmbH, Martin-Adolf Str. 44, 89165 Dietenheim, Germany.

Corresponding author:

Souvik Chakraborty, Department of Sustainability Technologies, Institute of Lightweight Systems, German Aerospace Center (DLR), Braunschweig, Germany.

Email: souvik.chakraborty@dlr.de

Data Availability Statement included at the end of the article

Mechanical recycling for instance is based on reducing the composites into granules (via crushing, shredding etc.), which are later used via injection moulding (as one of the options for thermoplastics) into new composites of the same type.³ Thermal recycling on the other hand includes thermal decomposition of scrap to generate energy.⁴ Although in some cases such as Fluidized Bed processing, fibers can be recycled in the manufacturing of new composites, but it is common that after processing at high temperatures, carbon fibers show up to 20% decline in stiffness.⁵ The commonly used thermal recycling method is pyrolysis, where polymer degradation results in liquid and/or gas with calorific value, which in some cases are similar to oils used as fuels.³ The high process temperature can simultaneously lead to a noticeable reduction in the intrinsic mechanical properties of the fibers as well.⁵ In comparison, chemical recycling degrades the polymeric matrix by dissolution using organic solvents (solvolysis) or water (hydrolysis), but it also generally requires higher pressure and temperatures (ranging from 200°C to 450°C) depending on the matrix targeted to be degraded. This results in fiber remnants and reactants containing monomers from resin, commonly low molecular weight oligomers that can possibly put to wider use (e.g., as fuels). Although the fiber recovery is successful with potentially comparable to virgin fibers, but the recovered oligomeric components are generally different from the matrix.⁴ This being said, the potential of recovering fiber and matrix components under milder conditions with limited devalorization is becoming increasingly possible with the gaining popularity of vitrimers.⁶

It is evident that the available recycling methods generally require processing conditions more intense than ambient as well as fall short to recover the matrix and fiber in a comparable quality to the virgin material which leads to inferior properties. These factors have a major influence on the cost of the recycling process, which makes it commercially intensive too.

Previous works from Gebhardt et al.^{7,8} presented two approaches for recycling Elium-based CFRPs: the first involved dissolving the composites to separate the fiber from the polymer, resulting in 81% matrix recovery rate with properties comparable to virgin material.⁷ The recycled fiber was then used to manufacture composites with very similar mechanical properties. The second approach involved reusing pre-polymerised Elium granulates from production waste (i.e., pre-polymerized recyclate) as an additive to virgin Elium monomer to manufacture CFRPs with reduced virgin material usage. This approach demonstrated that up to 7.5 wt% of virgin Elium can be saved to process CFRPs with final properties comparable to virgin counterparts.⁸

The goal of this work is to combine the two previous approaches^{7,8} and investigate the possibility of reusing recycled matrix (recovered recyclate) from dissolving Elium CFRPs in acetone, to reduce virgin matrix material usage for processing new 2nd generation composite parts. This can

lead to reducing waste whilst paving the path of a new life for recycled materials for this particular resin system, contributing to sustainable thermoplastic composite manufacturing.

Materials and methods

Materials

A bi-axial 0/90 carbon fiber scrims (width: 1270 mm, B-C-636gsm with polyester cross stitches of 6gsm) from Saertex was used to process the CFRPs with Elium® 150 from Arkema. The polymerization was completed using 2.5wt% of the initiator Perkadox CH-50X (dibenzoyl peroxide containing dicyclohexyl phthalate, with respect to the virgin resin used) from AkzoNobel by manually stirring till a pale-yellow mixture was obtained. Frekote 770NC from Henkel was used as the mould release agent. Reagent grade acetone from Henkel was used for recycling the Elium CFRPs to recover the matrix components.

Processing and recovery of Elium

Matrix recovery. Elium CFRPs were processed as has already been reported earlier^{7,8} and described briefly in Section S1 of the [Supplemental information](#). To mention briefly here, manufacturing via Vacuum-Assisted-Resin-Infusion (VARI) took place on a glass plate to enable the infusion process. The surface preparation was achieved by cleaning the plate with acetone, followed by coating the infusion designated area with Frekote. Two layers of the reinforcement were used to prepare the CFRP laminates. These as processed CFRPs were recycled via the reported process to separate the polymer from the fiber preform.⁷ Since the recovered Elium would be later dissolved in pure monomer, it was crushed (Dreher, Type S 26/26 GF) into granulates of size between 4 and 8 mm which, as already mentioned are designated as 'recovered recyclate'. The smaller granulates facilitate the dissolution in the Elium monomer as a consequence of a larger aspect ratio.⁹

Resin mixture preparation. Elium monomer was mixed with the recovered recyclate at the following loading fractions - 2.5wt.%, 5.0wt.% and 7.5wt.%. Each resin mixture weighed 100 g (including pure monomer, recyclate and the Perkadox CH-50X, at 2.5wt% of the virgin/new Eium resin used). The recovered recyclate were first added to the pure monomer with gradual stirring to aid in dissolution. The mixture then was left for a minimum of 24 h under ambient conditions, during which it was stirred for a second time to prevent agglomeration. However, the higher weight contents of 5.0% and 7.5% took up to 48 h to completely dissolve during colder months in comparison to warmer months. 2.5 wt% of the initiator was weighed in, in accordance to

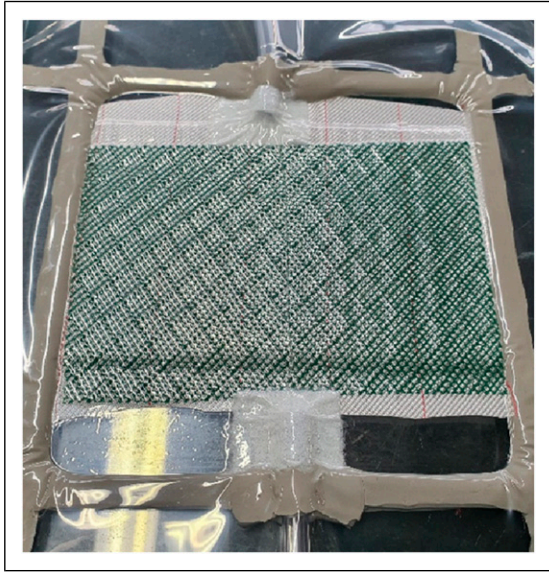


Figure 1. Laminate setup before infusion.

virgin resin directly before infusion. The viscosity noticeably increased further post initiator addition.

Composite processing. The mixture was VARI infused using two layers biaxial 0/90 CF preform with the expected target fiber volume fraction (V_f) of 45% approx. and dimensions of 100×150 mm in correlation to the previous work as shown in Figure 1^{7,8}. The 2 layers of biaxial fiber preforms were laid up symmetrically to limit any unwanted artefacts (e.g. warpage, residual stress buildup) leading to a final laminate thickness within 1.7–2.0 mm. This layup has been in congruence with our previous work for a uniform deduction of the measurements.

Prior to infusion, the resin was degassed to remove entrapped air bubbles that could affect resin homogeneity during the infusion process. Post-infusion the laminates were cured under a heat blanket at 65°C for 1.5 h.⁷

Characterization techniques

Differential scanning calorimetry (DSC). A heat flux DSC 2920 from TA Instruments was used to evaluate the subject material glass transition temperature (T_g , $^\circ\text{C}$) in compliance with DIN EN ISO 11,357-2. Samples were taken from the resin mixture after complete polymerization. A heat-cool-heat regime was performed under nitrogen atmosphere at a heating rate of $10^\circ\text{C}/\text{min}$ from -20 – 200°C , cooling down to -20°C at $50^\circ\text{C}/\text{min}$ and heating back up to 200°C with the ramp unchanged. The first heating provides information about the sample processing but, the second heating gives an insight into the structure of the polymer and the glass transition. Thus, the second DSC run was used to evaluate the T_g that is generally defined by the midpoint of step-change in the heat flow versus temperature plots.¹⁰

Thermogravimetric analysis (TGA). The measurements were performed to carry out a comparative evaluation of the thermal stability and decomposition behavior of the modified resin at different recovered recycle content from the threshold of 5 % weight loss known as onset of degradation (T_d). A TGA 2950 from TA Instruments was used, within a temperature range of 25°C to 600°C at a ramp of $10^\circ\text{C}/\text{min}$ under nitrogen atmosphere.

Gel permeation chromatograph (GPC). Gel permeation chromatography (GPC) measurements were used to assess the effect of addition of the recovered recycle on the molecular weight, and molecular weight distribution. Sampling was carried out by dissolving 8 mg of matrix granulates in tetrahydrofuran (THF) at 2 mg/ml and pumped with a flow rate of 1 ml/min at 25°C . In line with previous work, the same custom-made GPC unit was used that included 515 HPLC pump from Waters, an RI Detector 2300 from Smartline and a Marathon Autosampler from Knauer.⁸

Dynamic mechanical analysis (DMA). Considering that the focus of this analysis was to determine the glass transition range, DMA was carried out in the temperature sweep mode.¹¹ The thermomechanical analysis was performed according to DIN EN ISO 6721-4 with a DMA from METTLER. Specimens were machined from the infused laminates with dimensions of $80 \text{ mm} \times 10 \text{ mm}$. The test was conducted in a three-point bending mode (DMA 3PB) over a temperature range from -10°C to 160°C at a ramp rate of $2^\circ\text{C}/\text{min}$ and at a constant frequency of 1 Hz. The applied force ranged from 0.5 N to 1 N, with a time interval (dt) of 1.00 s. Synchronization was enabled.

Interlaminar shear strength (ILSS). ILSS was carried out in a Zwick/Roell BZ2-MM100 TL.ZW01 universal testing machine in accordance with DIN EN ISO 14,130 at 23°C and 45% humidity. The specimens were machined $20 \times 10.5 \text{ mm}^2$ and were loaded at a constant rate till a horizontal shear failure between the laminas occurred.¹² The span to thickness ratio were kept at 1:5 and was adapted to each specimen.

Scanning electron microscopy (SEM). Scanning Electron Microscopy (SEM) micrographs were captured to visualize the fiber-matrix interface to study the interfacial adhesion using a Helios G4 CX DualBeam system (FEI, USA) in ‘Field-Free’ mode at an accelerating voltage of 5 kV. A Leica EM ACE600 was used to sputter the fracture surface of the specimens from the 3PB tests with 4 nm Pt prior to imaging.

3-point bend (3PB) flexural measurements. Flexural tests in 3PB mode were carried out in compliance with DIN EN ISO 14,125, with 100 kN load cell and a preload of 5 N. Test



Figure 2. Pre-polymerised recyclate granulates (left) and recovered recyclate granulates (right).

specimens were machined according to the standard with dimensions $100 \times 15 \text{ mm}^2$. A thickness-to-span ratio of 1:16 was applied, where the mean thickness of the specimen was calculated, and the span length was adjusted accordingly.

Results and discussion

The process followed to obtain the recovered recyclate fraction, and its effect on the Elium resin mixture is discussed in detail here, starting with the visual assessment and the subsequent infusion behavior in comparison to the reference (virgin Elium-based system). This gave the chance to have an insight into the influence of the following factors individually.

- Visual appearance of the recovered recyclate
- Impact of the addition of recovered recyclate on the infusion characteristics of the modified resin
- Comparison with pre-polymerized recyclate reported earlier.⁸

Visual appearance of the recovered recyclate

The recovered recyclate fraction was obtained by dissolution of Elium CFRPs in acetone followed by drying, as it has been detailed previously.⁷ A clear visual distinction can be made from the pre-polymerized (crushed bulk plates as labelled in Ref. 8) recyclate, as can be observed in Figure 2. The recovered recyclate resemble flakes rather than particles, with a darker color. This can be a consequence of several factors, such as the amount and nature of the solvent used, as well as temperature and rate of evaporation affecting the solidification behavior of the polymer and consequently its visual appearance.

Impact on the infusion characteristics

One of the first aspects expected to be influenced by the addition of the recovered recyclate is the infusion behavior, especially with respect to the viscosity of the matrix and the polymerization rate. It has been observed that samples containing 2.5 wt% recovered recyclate showed successful infusion at a comparable time to the virgin material. However, the viscosity of the resin system modified with 5.0 wt% and 7.5 wt % of recovered recyclate has been observed to be high even prior to the initiator addition. The addition of the initiator then resulted in an expected further increase in viscosity, along with heat generation indicating the beginning of polymerization. This was especially evident at higher recovered recyclate

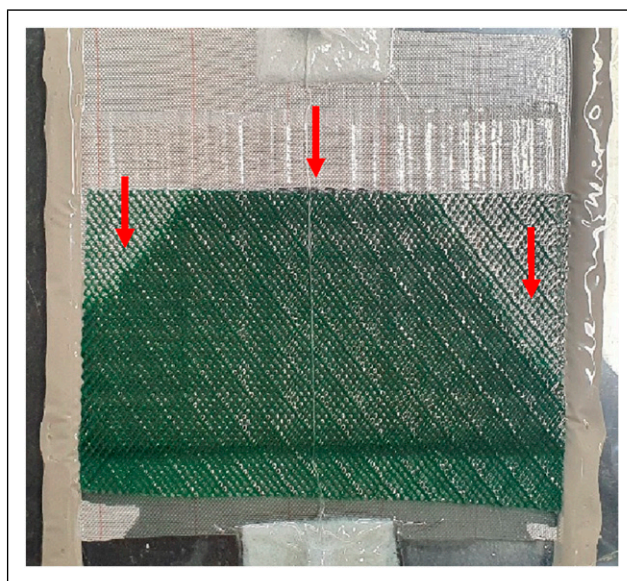


Figure 3. Infusion behavior of resin system modified with 7.5 wt% of recovered recyclate.

Table 1. GPC data of number average molecular weight, average molecular weight and dispersity, $\bar{D}$.

Entry	Recyclate wt%	Material/Type of recyclate	M_n (Da)	%Change M_n to virgin	M_w (Da)	%Change M_w to virgin	$\bar{D}$
1	0	Virgin Elium ⁶	70,700	-	293,000	-	4.12
2	100	Recycled Elium/heat recovered recyclate	62,490	-11.61	222,300	-24.13	3.56
3	2.5	Elum monomer polymerized with 2.5 wt% recovered recyclate	43,670	-38.23	105,500	-63.99	2.42
4	5.0	Elum monomer polymerized with 5.0 wt% recovered recyclate	46,630	-34.05	114,000	-61.09	2.44

content (7.5wt%), making the processing highly challenging (Figure 3) and resulting in incomplete infusion. However, the fact that the infusion is temperature dependent and it was carried out during different times of the year at room temperature, resulted in rare occasional complete infusion on warmer days even with 7.5wt% loading. This inconsistency led to exclude processing further laminates with resin modified with 7.5wt% of recovered recyclate.

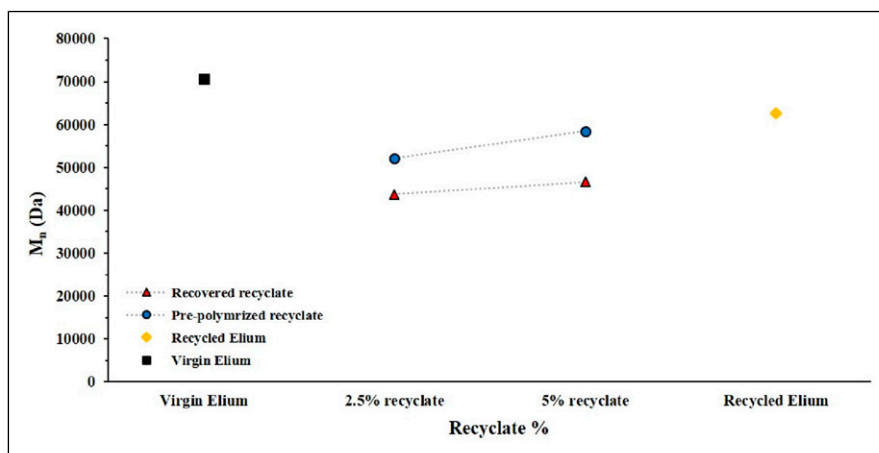
The increasing viscosity was also observed in previous work⁸ wherein complete infusion up to 5.0wt% of pre-polymerized granulates took a time similar to virgin Elium-based processing. However, with 7.5wt% the infusion time was 40% longer. The steep increase of viscosity after the addition of the initiator, can be attributed to the auto-acceleration or gel effect that causes instant increase of the polymerization rate.⁸ Since the termination is a diffusion-dependent process, once a high viscosity (namely the critical viscosity of the gel effect) is reached, the termination rate decreases, which in turn increases the polymerization rate. This phenomenon occurs in radical polymerization and leads in a sudden increase in the polymerization rate. Shi et al.¹³ reported that the dissolution of PMMA in MMA monomer reduced the time for the system to enter the auto-acceleration phase, which enhanced the polymerization rate, stating that the main influencing factors are the viscosity and the quality of the predissolved polymer.

A corresponding relation was also observed in this work between increasing the mass of the pre-dissolved polymer and the increase in the polymerization rate. This is due to the fact that the added recyclate fraction comprises inert higher molecular mass polymer network, and the progressively increasing recyclate content leads to a rise in the viscosity of the system. Once the initiator is added to the resin mixture, the system can reach the auto-acceleration phase earlier. This explains the slower infusion at higher recyclate content.

Effect on matrix

Evaluation of GPC measurements. To understand the effect of the recovered recyclate on the polymerization of the matrix, number average (M_n) and weight average molecular weight (M_w) as well as dispersity ($\bar{D}$) were evaluated.

When comparing to the virgin material, addition of recovered recyclate fractions in the polymerization mixture (Elum monomer + initiator) led to a decrease in both M_n and M_w , regardless of the loading fraction. A detailed examination of how varying concentrations of recyclates affect each parameter revealed that M_n exhibited a small increase from 2.5% (43.7 kDa) to 5.0% (46.6 kDa). Similarly, M_w also showed an analogous increase from 2.5% to 5.0%.

**Figure 4.** M_n of matrices with recovered vs. pre-polymerized recyclate.

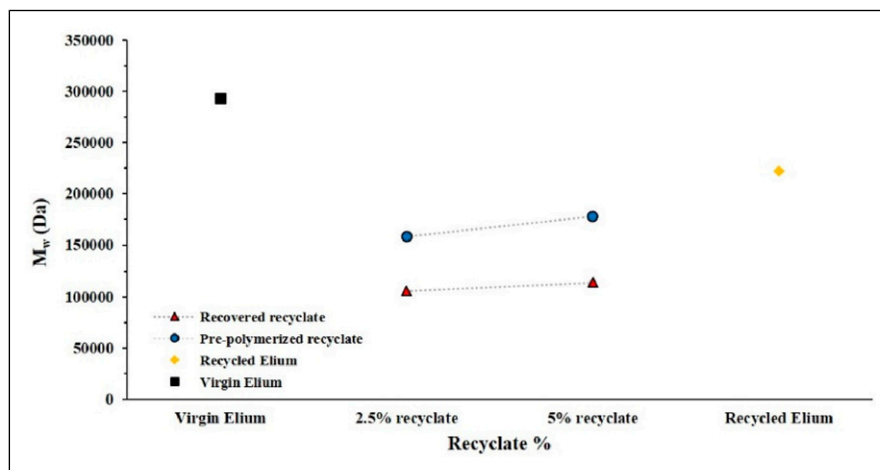


Figure 5. M_w of matrices with recovered vs. pre-polymerized recydate.

To understand whether this behavior is only due to the addition of a high molecular weight recydate fraction in the polymerization mixture or may depend also on the type of the recydate added, the data of Table 1 were compared to the equivalent data for Elium polymerization mixtures containing 2.5 wt% and 5.0 wt% of pre-polymerized recydate (pre-polymerized Elium granulates from production waste as mentioned in Ref. 8). The comparative data for M_n and M_w are given in Figures 4 and 5, respectively.

The integration of recydates (irrespective of the type) resulted in lower M_n and M_w compared to the virgin and recycled Elium reference materials, as shown in Figures 4 and 5 respectively. Ultimately, the inclusion of 5.0 wt% recydate resulted in a matrix with marginally higher M_n and M_w compared to the 2.5 wt% sample, again for both recydate types. Most notably, when comparing between the

same formulations in terms of percentage of added recydate, it is clear that the addition of pre-polymerised recydate results in higher molecular weight for both 2.5 and 5.0 wt% (blue data points in Figures 4 and 5) compared to the recovered recydate counterparts. This is expected as the pre-polymerised recydate is unused virgin Elium (Table 1, entry 1) which is of higher M_n , M_w and $\bar{D}$ compared to the recovered recydate additive, which is recycled Elium matrix from acetone extraction of Elium CFRPs (Table 1, entry 2).

The difference in molecular weight between the same wt% formulations with recovered and pre-polymerized recydates described above was further observed in the corresponding dispersity values. Similarly to our previous work,⁸ the addition of recovered recydate resulted in a narrower dispersity value for both 2.5 and 5.0 wt% formulations ($\bar{D} \approx 2.4$ for both), compared to both virgin and

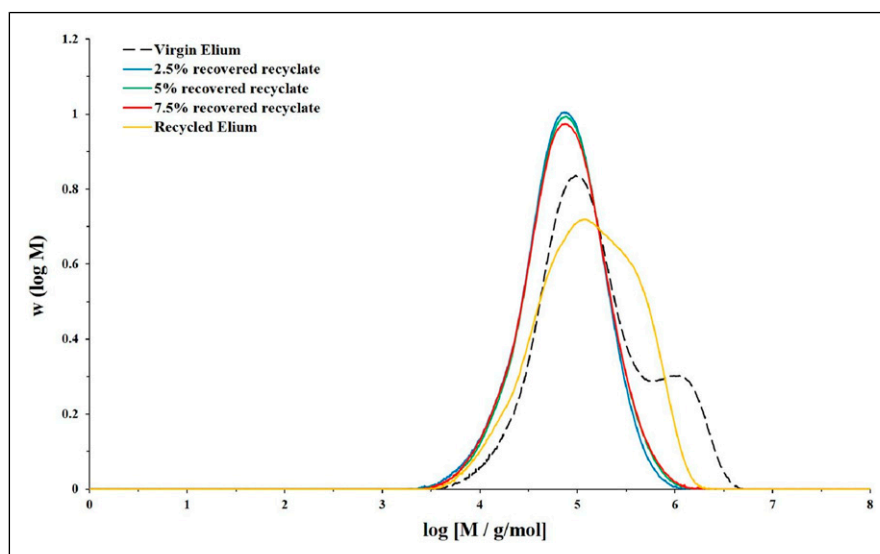


Figure 6. GPC measurements; Reference trace is virgin Elium (Table 1 entry 1).

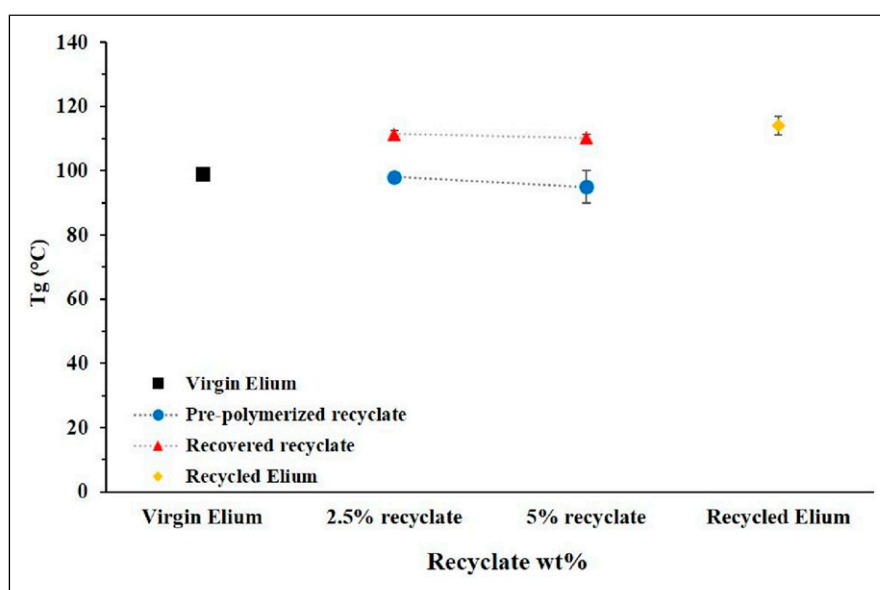
Table 2. Overview of thermal analysis of the modified resin systems – T_g from DSC and T_d from TGA.

Recyclate	T_g ($^{\circ}\text{C}$)				T_d ($^{\circ}\text{C}$)			
Material	0%	2.5%	5%	100%	0%	2.5%	5%	100%
Virgin Elium ⁷	99	-	-	-	215 (± 4)	-	-	-
Recycled Elium ⁷	-	-	-	114 (± 3)	-	-	-	190 (± 6)
Pre-polymerized recyclates ⁸	-	98 (± 1)	95 (± 5)	-	-	225 (± 8)	234 (± 3)	-
Recovered recyclate	-	111 (± 1)	110 (± 1)	-	-	249 (± 8)	251 (± 2)	-

recycled Elium (Table 1). This is attributed, as in the previous work,⁸ to a stereospecific templating effect^{14,15} during the polymerisation of the Elium monomer in the presence of the template-inducing high molecular weight recyclate. However, the type of recyclate appears to be a significant factor in this, with the lower molecular weight recyclate used here resulting in lower $\bar{D}$ values compared to the pre-polymerized recyclate additives ($\bar{D} \approx 3.0$ for both 2.5 and 5.0 wt% formulations⁸). This in turn is attributed to the discussed difference in recyclate molecular weight, with the lower molecular weight recovered recyclate resulting in a more uniform molecular weight distribution and lower $\bar{D}$ of the formed network. This is a particularly interesting feature to explore further, as even lower M_w recyclate additives could lead to formulated matrix networks with even more uniform molecular weight distribution and hence tailored mechanical and thermo-mechanical performance. The relevant molecular weight distribution curves from GPC measurements are given in Figure 6, illustrating the progressively more unimodal

molecular weight distribution profile from virgin to recycled and ultimately the two formulated 2.5 and 5.0 wt% products.

Differential scanning calorimetry (DSC). The impact of the addition of the recovered recyclate on the thermal properties has been investigated by DSC and TGA. The overall results have been tabulated in Table 2. As a general observation, addition of recyclates appears to slightly increase the T_g compared to virgin Elium. The recovered recyclate appears to induce a comparatively higher increase (11 – 12 $^{\circ}\text{C}$) to the pre-polymerized recyclates, where the T_g was rather unchanged. However, even the observed higher T_g increase for the formulations with the recovered recyclate is within an expected variation margin of up to 10 $^{\circ}\text{C}$ for high molecular weight polymer networks, therefore it would be overall considered at least unchanged rather than consistently increased. Moreover, from Figure 7 it becomes clear that the T_g remains fairly constant irrespective of the recyclate content within the same recyclate class.

**Figure 7.** T_g -midpoint of matrices with recovered vs. pre-polymerized recyclate.

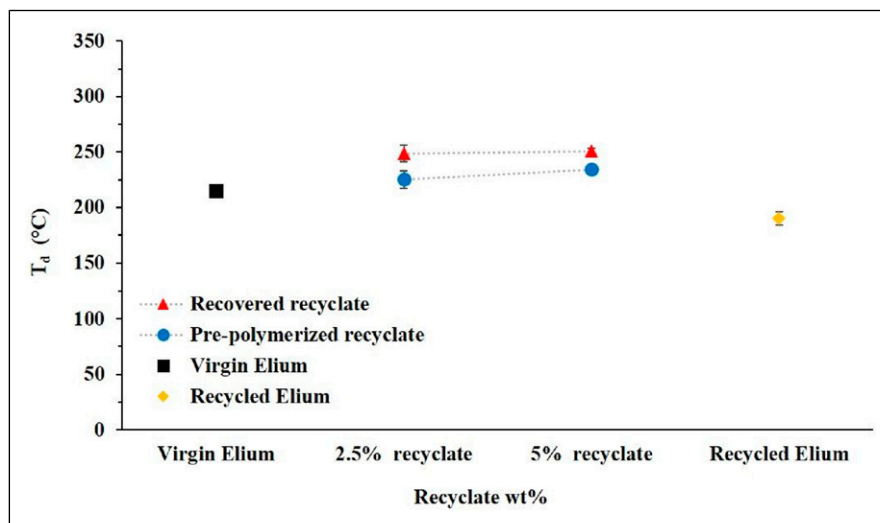


Figure 8. T_d of matrices with recovered vs. pre-polymerized recyclate.

Thermogravimetric analysis (TGA). To assess the impact of the addition of the recovered recyclate on the thermal degradation of the modified matrix, TGA was carried out to obtain the onset degradation temperature (T_d) value which generally corresponds to 5% weight loss. Comparison with the samples containing pre-polymerized recyclates⁸ was conducted, as before for molecular weight and T_g . The overview of the TGA measurements is presented in Table 2, Figure 8 and Figure S1 of supplemental information. As a general observation, the resin modified with recovered recyclate exhibited a higher T_d (irrespective of the recyclate content), indicating improved thermal stability. In addition, the degradation profile (Figure S2) for the resins with recovered recyclate were comparable.

As can be seen in Figure 8, with increase in recyclate content a corresponding increase of T_d has been observed, with recovered recyclate formulations exhibiting greater increase compared to pre-polymerized counterparts. Moreover, increasing the recyclate content to 5.0 wt% resulted in a higher degree of thermal stability of both modified matrix types. In line with our previous study,⁸ the increase in T_d between each recyclate class with recyclate

content is attributed to the increasing M_n of the respective formulation (from 2.5 to 5.0 wt%).

Summarising the thermal analysis studies, the addition of recovered recyclate fractions in the Elium resin resulted in at least comparable (T_g) and in some instances improved (T_d) thermal properties of the corresponding formulations in relation to virgin Elium. No detrimental effect on thermal properties or stability was observed from the addition of recovered recyclate, in line with the pre-polymerized counterparts.

Dynamic mechanical analysis (DMA). Dynamic mechanical analysis was employed to study the thermomechanical performance of the manufactured laminates along with a first indication of the compatibility between the CF pre-forms and resin modified with both pre-polymerised and recovered recyclate fractions. For comparison, the assessment process followed is in line with the previously reported work.⁸ The glass transition temperature values from both $T_{g \text{ onset}}$ (from E') and $T_{g \text{ tan } \delta \text{ max}}$ along with the storage modulus have been used to assess the fiber-matrix compatibility. Finally, the damping performance was probed by assessing

Table 3. Overview of DMA results and how it compares with Ref. 7.

Material	Recyclate content [wt%]	T_g (Onset) (°C)	E' (MPa)	Integral Tan δ 80-140°C (°C)
Virgin Elium ⁷	0.0	88.7 (± 4)	8952 (± 969)	16.7 (± 1.1)
Pre-polymerized recyclate ⁸	2.5	86.8 (± 1)	13065 (± 3510)	17.1 (± 1.2)
	5.0	84.0 (± 3)	8976 (± 759)	16.4 (± 0.3)
Recovered recyclate	2.5	93.7 (± 4)	17381 (± 372)	12.6 (± 0.3)
	5.0	97.8 (± 0.7)	11786 (± 1801)	14.0 (± 0.8)

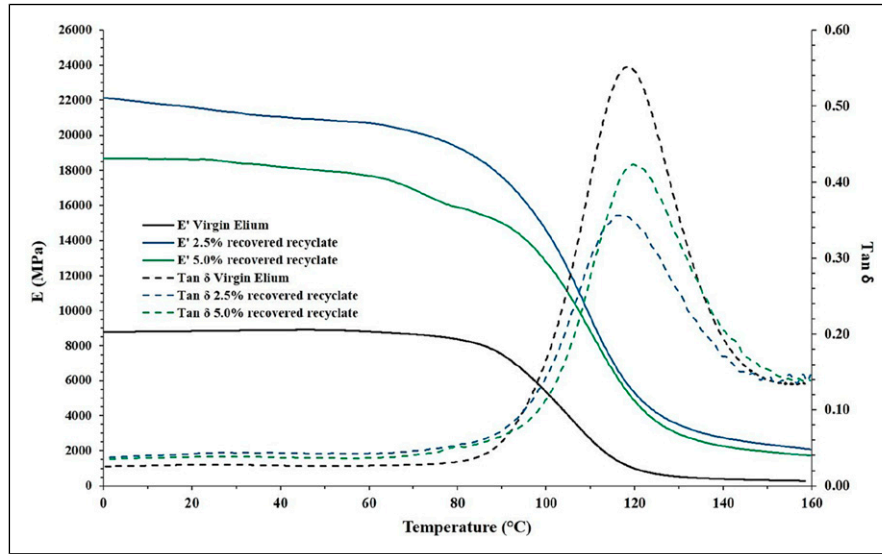


Figure 9. DMA comparison with virgin material.

the integral of the $\tan \delta$ curve of the specimens. The results are summarised in Table 3 and Figure 9.

From Table 3, it becomes clear that the addition of the recovered recyclate fraction to modify the virgin resin system has a clear influence on the onset T_g value. The improvement is better than previously observed with the pre-polymerized recyclates.⁸ Laminates containing 5 wt% recovered recyclate exhibited the highest $T_{g \text{ onset}}$, followed by the 2.5 wt%. In comparison, the matrix modified with the pre-polymerized Elium granulates exhibited the lowest $T_{g \text{ onset}}$. This complements the DSC measurements by confirming that the substitution of monomer with recyclates influences the glass transition temperature.

Simultaneously, laminates processed with Elium modified with recovered recyclate displayed a noticeable improvement in storage modulus with 2.5 wt% (+94%) and 5.0 wt% (+32%) in comparison to the virgin Elium reference system. Higher storage modulus can be attributed to improved uniformity of the matrix (as represented by the lower dispersity values and hence narrower molecular weight distribution, Table 1 and Figure 6), as well as to improved fiber-matrix compatibility.

Furthermore, the shape of the loss factor ($\tan \delta$) curve has been studied as the characteristic measure of the damping of the composite specimens, which also aids in understanding the fiber-matrix compatibility.¹⁶ Mechanical damping of polymers is their ability to dissipate energy through internal friction during deformation. In a simplified manner, a reduction in damping (reduced $\tan \delta$ integral) is indicative of improved fiber-matrix compatibility which is often reflected in enhanced bulk properties. This was evaluated by integrating the area under the $\tan \delta$ curve between 80°C and 140°C. As it can be seen in Figure 9, composites processed with resin modified with recovered recyclate (2.5-5.0 wt%)

showed lower peak areas in comparison to the reference and to composites processed with resins modified with pre-polymerized granulates.⁸ From the above analysis and low damping values, one can expect an improvement in the bulk composite properties.^{8,17}

Effect on composite properties

Visualizing the interface. To further investigate the perceived improvement in the fiber-matrix compatibility suggested by the DMA analysis and the reduced $\tan \delta$ values, SEM images of laminate fracture surfaces were obtained and are shown in Figure 10.

As a general observation, the laminates processed with resin modified with the recovered recyclate showed strong interfacial adhesion (no fiber pull-out observed, Figure 10(a) and (d)) irrespective of the recyclate content. Interestingly, the surface of the laminates processed with 5.0 wt% recovered recyclate seems to be rougher (more granular, Figure 10(d) and (e)) than that with 2.5 wt% (Figure 10(c)). Complementarily, the signatures of strongly embedded fibers (white arrows in Figure 10(b) and (e)) along with the matrix wrapped around fibers (red arrow Figure 10(c)) and riverbed-like patterns (Figure 10(f)) are further evidence that the use of recovered recyclate to modify the virgin resin has no detrimental effect on the fiber-matrix interfacial adhesion.¹⁸ These were prevalent in the virgin fiber based Elium CFRPs as well (please refer to Figure S3 in the supplementary information).

These observations are comparable to those made in the previous work with virgin Elium CFRPs as well as with the resin modified with pre-polymerized recyclates.^{7,8} A comparative micrographic analysis is given in the supplementary information (Figure S3). Based on the above, it can be

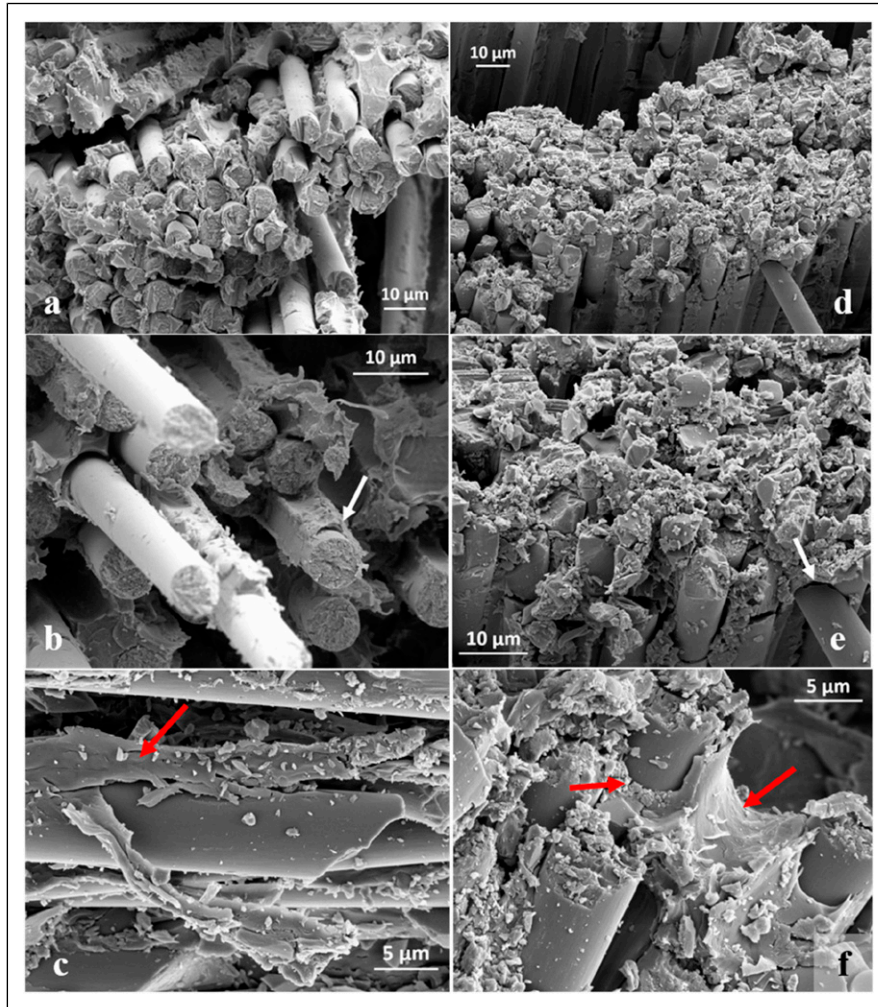


Figure 10. SEM micrographs, left (a–c) for resin modified with 2.5wt% of the recovered recyclate and right (d–f) for those modified with 5.0 wt% recovered recyclate.

expected that the bulk performance of the corresponding laminates would not significantly deteriorate in comparison to the virgin Elium-based specimens.

Interlaminar Shear stress measurements (ILSS). For a first assessment of the observed fiber-matrix compatibility (from

Figure 10), the specimens have been subjected to ILSS tests to evaluate this. The interface being crucial in affecting the load transfer between the matrix and the fibers which eventually impacts the bulk structural properties. The results have been summarized in Table 4 of the Supplemental Information and shown in Figure 11 below. As observed,

Table 4. Overview of bulk mechanical testing.

Material	Recyclate content (wt%)	Flexural strength	Bending modulus	Shear stress at break
		(MPa)	(GPa)	(MPa)
Virgin Elium ⁷	0.0	480 (±48)	12.5 (±2.5)	31.50 (±5.3)
Pre-polymerized recyclate ⁸	2.5	496 (±83)	13.4 (±1.5)	36.69 (±4.7) ^a
	5.0	501 (±51)	13.4 (±1.3)	34.37 (±5.2) ^a
Recovered recyclate	2.5	290 (±107)	1.3 (±0.21)	45.68 (±15.0)
	5.0	352 (±55)	5.9 (±20.50)	33.46 (±9.1)

^aFrom un-published data as part of the test campaign for Ref. 8.

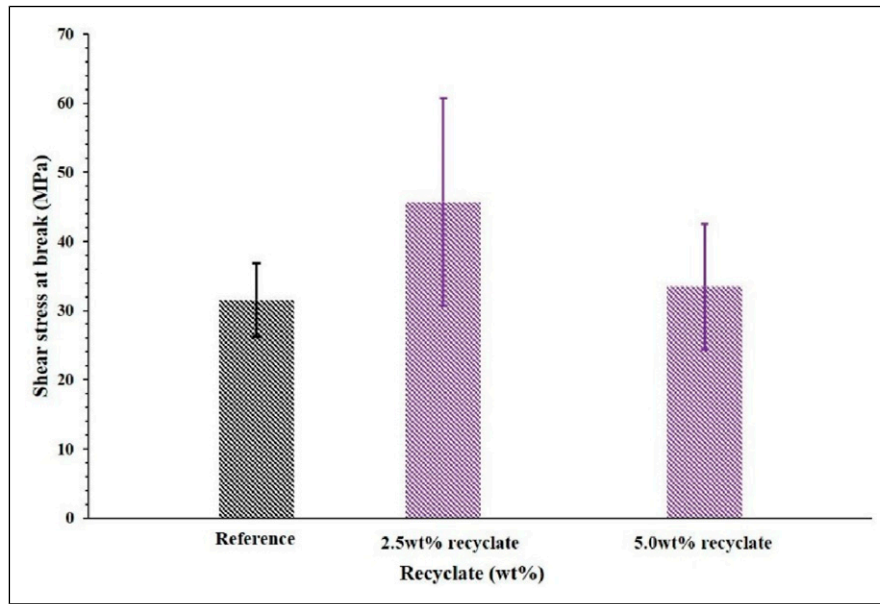


Figure 11. Shear stress at break from ILSS measurements for composites processed with virgin Elum reference and its comparison to resins modified with recovered recyclate.

it shows an improvement in the mean shear stress with the modified resins in comparison to the virgin counterpart. On the other hand, a decrease in mean stress is observed for resins modified with 5wt% recovered recyclates (in comparison to 2.5wt%) but this is still comparatively higher (although marginally) to the reference.

High interlaminar shear strength is often attributed to good bonding and stronger fiber-matrix compatibility which is complementary to the fractographic observations. Translation of these interfacial observation to structural tests offered a different observation, the details of which are

available in the following Section 2.4.3. However, this being said, the observed high standard deviation of ILSS measurements completely ignored (often time making it challenging to explain) as has been discussed by Graupner and Müssing¹⁹ is likely to be a consequence of failure modes other than pure shear (e.g. tensile, compression, etc.) acting simultaneously. This is only amplified by the more complex interface in the specimens itself.

3-point bend (3PB) flexural measurements. Given the encouraging observations from visualizing the fiber-matrix

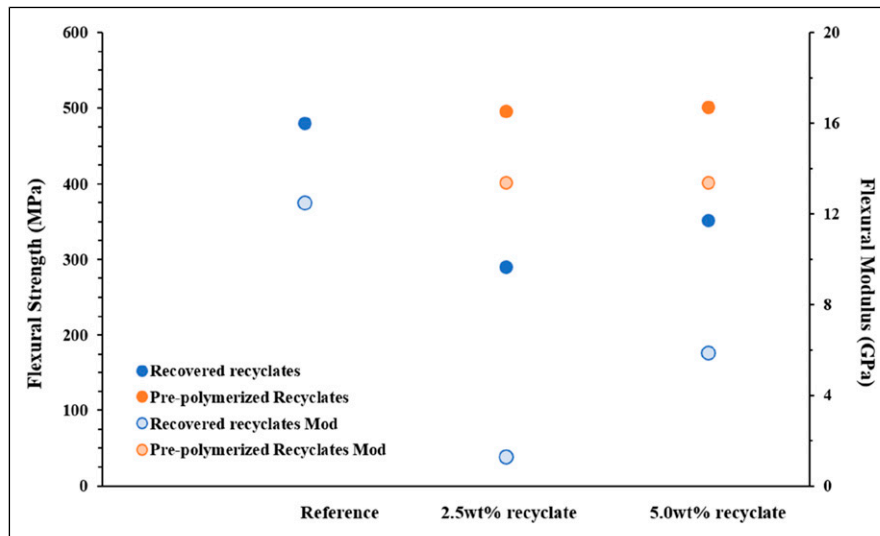


Figure 12. Flexural properties of the composites processed using resin modified with the recovered recyclate as compared to the recovered recyclate.⁸

interface (Figure 10) and the ILSS measurements (Figure 11), the impact on the bulk mechanical response was tested via 3-point bend flexural measurements. These tests would ideally shed some light on the stiffness (i.e., modulus) and on the load bearing capability of the material under bending (i.e., strength). The summary of these tests is presented in Table 4 and Figure 12.

Based on previous observations of improved interfacial compatibility (as shown by SEM and ILSS measurements), the strength and moduli values of the composites processed with resin modified with recovered recyclate were observed to be lower than those of the composites processed with resin modified with pre-polymerized recyclate. The reason for this is still not clear and a point for further investigation as both SEM and ILSS analysis suggested good interfacial compatibility. A lower stiffness is an indicator for improved flexibility and ductility but an answer to the overall reduced value needs more detailed studies.

Conclusion

This research demonstrates the potential of using recovered Elium recyclate as additives in the manufacturing process of new CFRP. The employed approach can produce composites with properties comparable to those manufactured with virgin Elium. A maximum of 5.0 wt% of recovered recyclate can be used as replacement of virgin Elium, resulting in composite laminates with acceptable mechanical properties and improved thermal stability. However, only the infusion of up to 2.5% is regarded feasible, as by 5 % the high viscosity may cause problems on a larger scale.

Although GPC showed a decrease in molecular mass and molecular number of the resin modified with recovered recyclate, the recyclate addition proved to be beneficial in narrowing the molecular weight distribution. The enhanced thermal stability and interfacial properties can also be attributed to the good adhesion between fiber and matrix, as supported by DMA, ILSS and SEM measurements.

In conclusion, the inclusion of up to 5 wt% recovered recyclate leads to higher thermal stability and improved interfacial properties, demonstrating a further substitution opportunity of reducing raw material by recycled Elium. While infusion remained feasible with 2.5 wt% recovered recyclate by room temperature, at 5.0 wt%, the increased viscosity resulted in potential processing challenges, indicating the need for process optimization (like in RTM, press, etc.) when using higher concentrations of recovered recyclate. Although the substitution seems to be a fraction in the present scale, when scaled up to the wider applicability of Elium in the recent times (e.g. in manufacturing WT blades), this is likely substitute the virgin resin by some significant amount. Additionally, it has to be taken into account that this process demonstrates the reuse of recycled and recovered material as a substitute for virgin material which in itself is significant step towards sustainability. The

inclusion of the waste into the virgin resin happens at room temperature and in the container used for processing the composite laminates, which eliminates the need to both energy and specialized equipment that speaks for the simplicity of the process itself.

ORCID iDs

Steffen Opitz  <https://orcid.org/0000-0003-1257-547X>

Souvik Chakraborty  <https://orcid.org/0000-0001-5217-4035>

Funding

The authors received no financial support for the research, authorship, and/or publication of this article.

Declaration of conflicting interests

The authors declared no potential conflicts of interest with respect to the research, authorship, and/or publication of this article.

Data Availability Statement

Data sharing not applicable to this article as no datasets were generated or analyzed during the current study.

Supplemental material

Supplemental material for this article is available online.

References

1. *Grand View Research: Thermoplastic Composites Market Size, Share & trends analysis report by end-use (transportation, aerospace & defense), by region, and segment forecasts, 2022–2030.*
2. Mallick PK. *Fiber-reinforced composites – materials, manufacturing, and design.* CRC Press, 2008.
3. Qureshi J. A review of recycling methods for fiber reinforced polymer composites. *Sustainability* 2022; 14: 16855, Heft 24, S. <https://doi.org/10.3390/su142416855>
4. van Oudheusden AA. *Recycling of composite materials.* Springer, 2019.
5. Asmatulu E, Twomey J and Overcash M. Recycling of fiber-reinforced composites and direct structural composite recycling concept. *J Compos Mater* 2014; 48: 593–608, Heft 5, S. <https://doi.org/10.1177/0021998313476325>
6. Hao C, Zhao B, Shao L, et al. Mild chemical recycling of carbon fiber-reinforced epoxy composites in aqueous buffers and development of hydrothermally recyclable vitrimer composites from recyclates. *Resour Conserv Recycl* 2024; 207: 10766. <https://doi.org/10.1016/j.resconrec.2024.107668>
7. Gebhardt M, Chakraborty S, Manolakis I, et al. Closed-loop room temperature recycling of Elium CFRPs and its influence on the 2nd generation composite properties. *Zeitschrift Kunststoffechnik* 2020; 1: S.179–210. <https://doi.org/10.3139/O999.02052020>
8. Gebhardt M, Manolakis I, Chatterjee A, et al. Reducing the raw material usage for room temperature infusible and

- polymerisable thermoplastic CFRPs through reuse of recycled waste matrix material. *Compos B Eng* 2021; 216: 108877. <https://doi.org/10.1016/j.compositesb.2021.108877>
9. Seager RJ, Acevedo AJ, Spill F, et al. Solid dissolution in a fluid solvent is characterized by the interplay of surface area-dependent diffusion and physical fragmentation. *Sci Rep* 2018; 8(1): 7711. <https://doi.org/10.1038/s41598-018-25821-x>
 10. Wunderlich B. *Thermal analysis of polymeric materials*. Springer, 2005.
 11. Frick A, Stern C and Muralidharan V. *Practical testing and evaluation of plastics*. Wiley-VCH Verlag GmbH & Co, 2019.
 12. *ISO 14125 Fiber-reinforced plastic composites-Determination of flexural properties*. ISA, 2017.
 13. Shi H, Zhuang Q, Zheng A, et al. Study of the radical polymerization mechanism and its application in the preparation of high-performance PMMA by reactive extrusion. *RSC Adv* 2023; 13(11): 7225–7236. <https://doi.org/10.1039/D2RA06441C>
 14. Połowiński S. Template polymerisation and co-polymerisation. *Prog Polym Sci* 2002; 27: 537–577, Heft 3, S. [https://doi.org/10.1016/S0079-6700\(01\)00035-1](https://doi.org/10.1016/S0079-6700(01)00035-1)
 15. Gons J, Vorenkamp EJ and Challa G. Radical polymerization of methyl methacrylate in the presence of stereoregular poly(methyl methacrylate). V. Kinetics of initial template polymerization. *J Polym Sci Polym Chem Ed* 1975; 13: 1699–1709, Heft 7, S. <https://doi.org/10.1002/pol.1975.170130720>
 16. Gedde UW and Hedenqvist MS. *Fundamental polymer science, Graduate texts in physics*. Springer, 2019.
 17. Gerard JF, Perret P and Chabert B. Viscoelastic study of carbon/epoxy unidirectional composite materials. *Adv Mater Sci* 1989; 23(1): 139–147. <https://doi.org/10.1002/masy.19890230111>
 18. Stoeffler K, Andjelic S, Legros N, et al. Polyphenylene sulphide (PPS) composites reinforced with recycled carbon fiber. *Compos Sci Technol* 2013; 84: 68–71.
 19. Graupner N and Müssig J. Interfacial and interlaminar shear strength of unidirectional viscose fibre-reinforced epoxy Composites—An overview of the comparability of results obtained by different test methods. *Front Mater* 2022; 9: 709845. <https://doi.org/10.3389/fmats.2022.709845>