Dependence of SEI growth on state-ofcharge and cycling conditions

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Electrochemical Energy Storage



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MOTIVATION AND AIM

- The **SEI and SEI growth** is crucial but not fully understood.
- The relevant **transport mechanisms are** still under debate [1]. •
- **Electron diffusion** vs. solvent diffusion as growth mechanism. •
- Coupling of chemical, electrical, and mechanical effects.

COMPARISON WITH SOLVENT DIFFUSION

Theory of SEI growth

• Reaction and diffusion: $\partial_t \tilde{L}_{\text{SEI}} = \frac{v j_0}{FL_{\text{SEI},0}} \cdot \frac{e^{-(1-\alpha)\tilde{\boldsymbol{U}}_0} - e^{\alpha \tilde{\boldsymbol{U}}_0} - \tilde{\boldsymbol{U}}_{\text{SEI}}}{1 + \frac{j_0 L_{\text{SEI},0}}{FDc_{\text{EC},\infty}}} e^{-(1-\alpha)\tilde{\boldsymbol{U}}_0} \tilde{L}_{\text{SEI}}$

SOC dependence of calendar aging

SEI GROWTH DURING BATTERY STORAGE [2]

Mechanisms in the literature

- Solvent diffusion
- Electron tunneling
- Electron conduction
- Electron diffusion through localized states





- **Electron / Li interstitial diffusion**
- Validation with capacity fade during calendar aging
- **Explains SOC dependence**

Migration

Other mechanisms fail

SEI GROWTH DURING BATTERY CYCLING

 \succ Either SOC dependence or $t^{0.5}$ reproduced



SEI MECHANICS [7]

- Silicon particle deforms chemically \mathbf{F}_{ch} and elastically \mathbf{F}_{el}
- **SEI** deforms plastically \mathbf{F}_{pl} and elastically \mathbf{F}_{el} , SEI failure



 10^{-4}

best fi

Mechanics and chemistry coupled via free energy density

 $\psi = c_{\text{Li,max}} \left[\mu_{\text{Li}}^0 \tilde{c}_{\text{Li}} + RT(\tilde{c}_{\text{Li}} \ln \tilde{c}_{\text{Li}} + (1 - \tilde{c}_{\text{Li}}) \ln(1 - \tilde{c}_{\text{Li}})) \right] + \frac{1}{2} \mathbf{E}_{\text{el}} : \mathbb{C} \mathbf{E}_{\text{el}}$





Growth regimes [4]

- If $L_{\rm app} \ll L_{\rm diff}$ $L_{\rm SEI} = \frac{v_{\rm SEI}}{E} j_{\rm SEI,0} e^{-\alpha_{\rm SEI} \tilde{\eta}_{\rm SEI}} \cdot t$
- If $L_{
 m diff} \ll L_{
 m app} \ll L_{
 m mig}$

$$L_{\text{SEI}} = L_{\text{tun}} + \sqrt{2\nu_{\text{SEI}}c_{\text{Li},0}De^{-\tilde{\eta}_{\text{SEI}}} \cdot t} + \left(L_{\text{SEI},0} - L_{\text{tun}}\right)^2$$

- If $L_{
 m mig} \ll L_{
 m app}$
 - $L_{\rm SEI} = \frac{v_{\rm SEI} c_{\rm Li,0} DF j_{\rm int}}{2RT \kappa_{\rm Li^+,SEI}} e^{-\tilde{\eta}_{\rm SEI}} \cdot t$



Validation with automotive cells [5]

Ltun

- Excellent agreement at moderate temperature
- Additional degradation mechanism at low temperature

Inhomogeneous SEI growth



SUMMARY

- **Electron diffusion** through localized states can explain **SOC** dependence and heterogeneous SEI growth in electrodes.
- Different **SEI growth regimes** during battery cycling validated with experimental data.
- Solvent diffusion can not describe the SOC dependence and $t^{0.5}$
- SEI deformation and cracking determines capacity fade of silicon electrodes.

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