

# Microstructure-sensitive modelling of deformation and fracture of TiAl alloys

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DLR Köln  
05.12.2017



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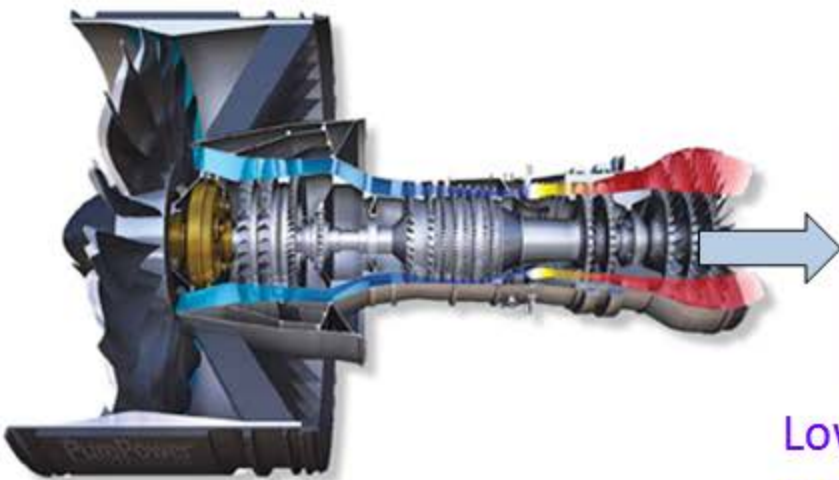
Citation:  
MR Kabir, Werkstoff-Kolloquium 2017, Institute of Materials Research,  
German Aerospace Center, Cologne, Germany.



Knowledge for Tomorrow



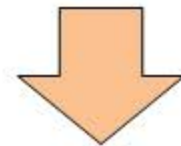
# TiAl for turbine blade: Requirement and goal



Low pressure turbine blade

## Need for improvements in mechanical Properties:

- Strength at high temperature
- Creep resistance
- Ductility
- Fatigue strength
- Fracture toughness




Can be improved by



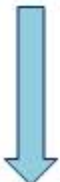
## Outline:

# Modelling of TiAl alloy behaviour considering microstructure sensitivity

Three problem areas will be demonstrated



Modelling of  
deformation  
behaviour



Modelling of crack  
initiation and  
propagation

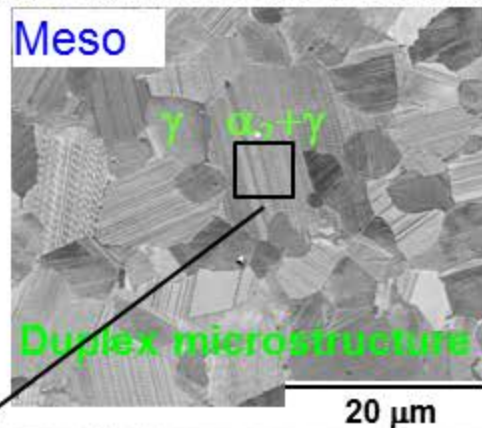


Modelling of dynamic  
fracture

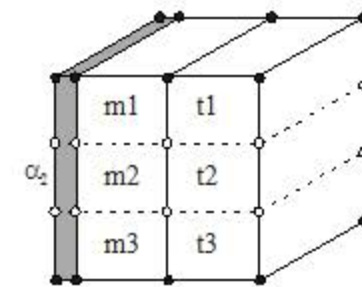
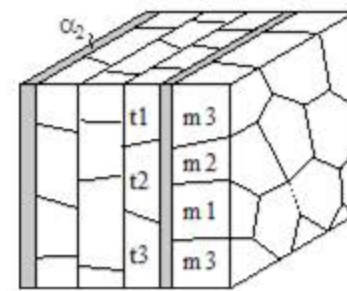
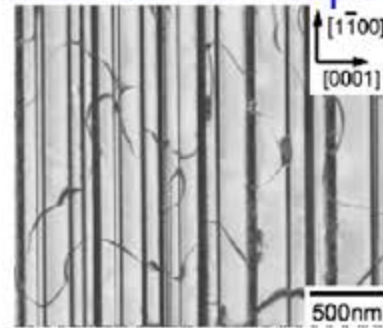


# Micromechanical modelling: Deformation of multi-phase structure

Kabir et al., Mat Sci Eng A 635(2015)13-22

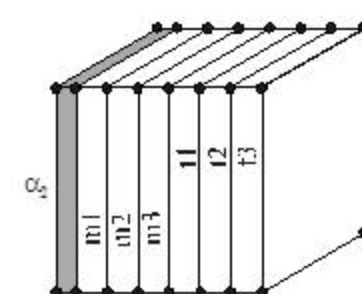
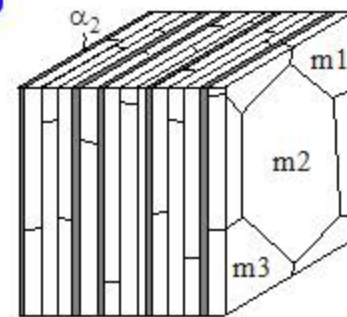
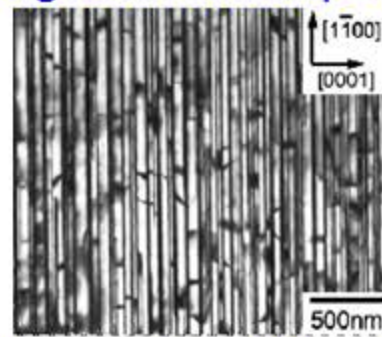


## Small domain aspect ratio



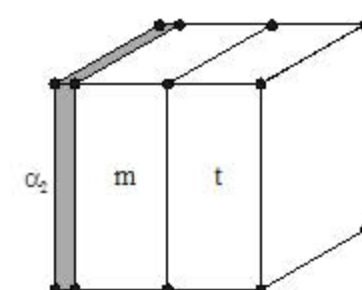
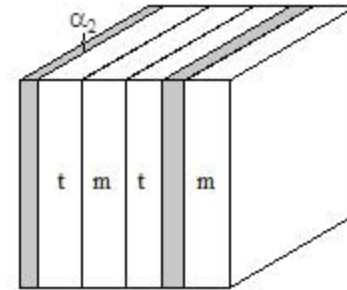
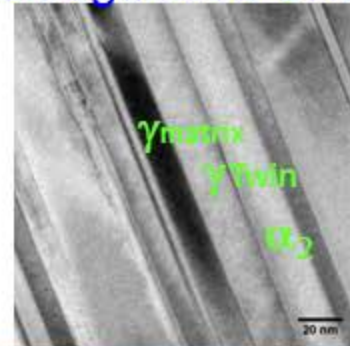
Equal-strain  
(Voigt) model

## Large domain aspect ratio



Equal-stress  
(Reuss) model

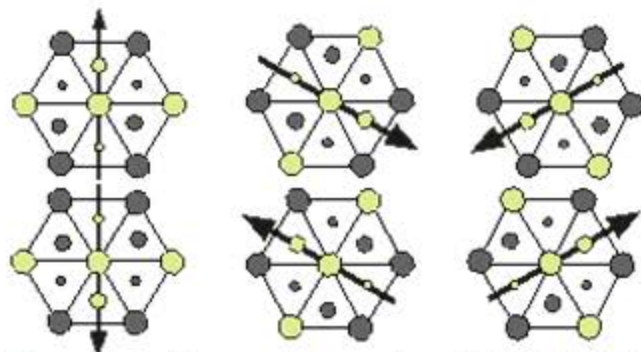
## Homogenized domain



Homogeneous  
lamellae

Werwer et al., Int J Plast 22 (2006) 1683

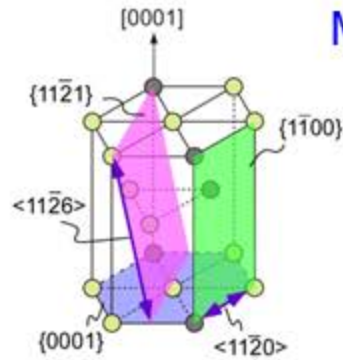
Matrix orientations of domains: 0°-120°-240°



Twin orientations of domains: 180°-300°-60°

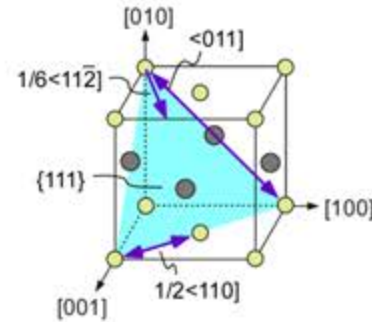
# Micromechanical modelling: Deformation of multi-phase structure

## Major Phases



$\alpha_2 \text{Ti}_3\text{Al}$

Prismatic  $\langle 11\bar{2}0 \rangle \{1100\}$   
 Basal  $\langle 11\bar{2}0 \rangle \{0001\}$   
 Pyramidal  $\langle 11\bar{2}6 \rangle \{11\bar{2}1\}$



$\gamma \text{TiAl}$

Ordinary  $1/2 \langle -110 \rangle \{111\}$   
 Twinning  $1/6 \langle -112 \rangle \{111\}$   
 Super  $\langle -110 \rangle \{111\}$

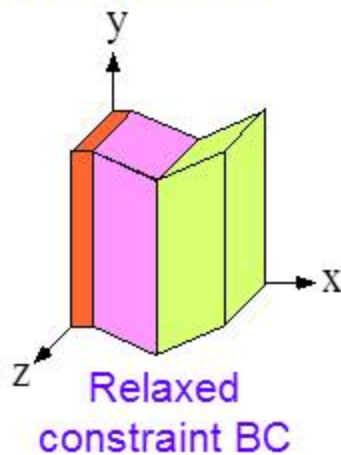
### Slip systems in $\gamma$ -TiAl

Slip type			
Ordinary	$1/2 [1\bar{1}0] \{111\}$	$1/2 [1\bar{1}0] \{11\bar{1}\}$	$1/2 [110] \{1\bar{1}\bar{1}\}$ $1/2 [110] \{1\bar{1}\bar{1}\}$
Super	$[0\bar{1}\bar{1}] \{111\}$ $[10\bar{1}] \{111\}$	$[0\bar{1}\bar{1}] \{1\bar{1}\bar{1}\}$ $[10\bar{1}] \{1\bar{1}\bar{1}\}$	$[0\bar{1}\bar{1}] \{11\bar{1}\}$ $[0\bar{1}\bar{1}] \{1\bar{1}\bar{1}\}$ $[\bar{1}0\bar{1}] \{11\bar{1}\}$ $[\bar{1}0\bar{1}] \{1\bar{1}\bar{1}\}$
Twinning	$1/6 [1\bar{1}\bar{2}] \{111\}$	---	$1/6 [\bar{1}\bar{1}\bar{2}] \{11\bar{1}\}$ $1/6 [\bar{1}\bar{1}\bar{2}] \{1\bar{1}\bar{1}\}$ $1/6 [1\bar{1}\bar{2}] \{11\bar{1}\}$

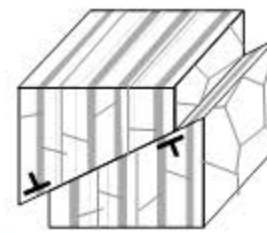
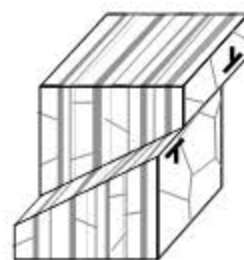
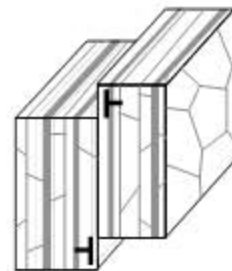
## Deformation of lamellar phases



Appel, F. (2005)  
*Philos. Mag.*, **85**, 205



## Deformation modes for lamellar grains



Werwer et al., *Int J Plast* 22 (2006) 1683

### Slip systems in $\alpha_2$ ( $\text{Ti}_3\text{Al}$ )

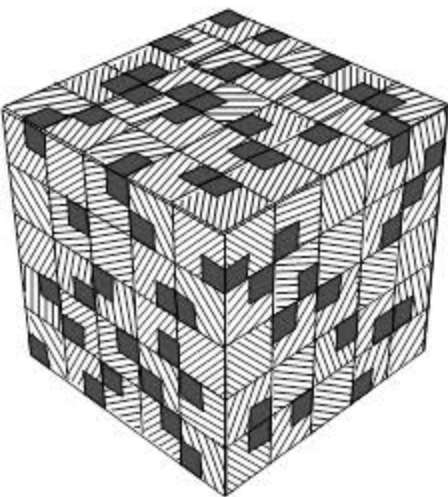
Slip type			
Prismatic	---	$\langle 11\bar{2}0 \rangle \{1\bar{1}00\}$	---
Basal	$\langle 11\bar{2}0 \rangle \{0001\}$	---	---
Pyramidal	---	---	$\langle \bar{1}\bar{1}\bar{2}6 \rangle \{11\bar{2}1\}$

Lebensohn et al., *Acta mater* 46 (1998)4701

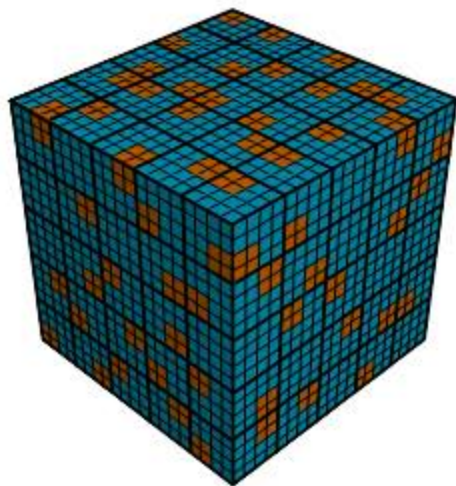


# Polycrystal models for FE analysis

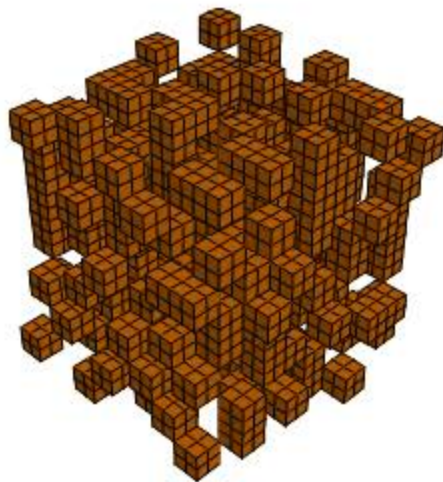
Voxel based polycrystal model



Idealization of the microstructure

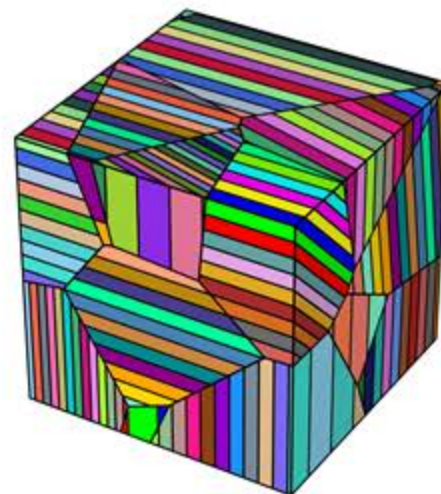
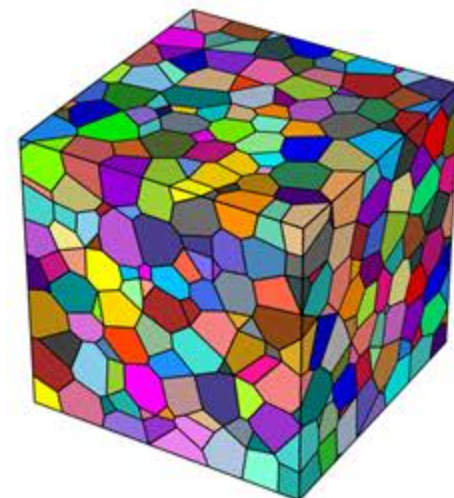
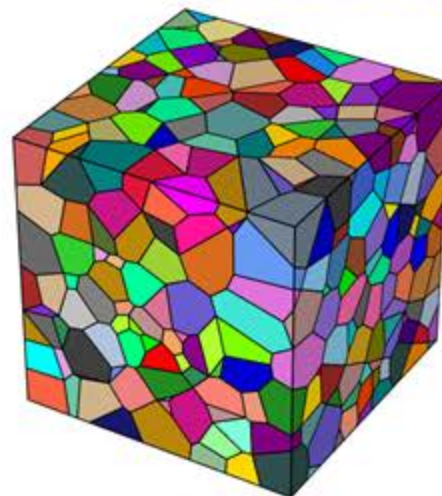


FE Model

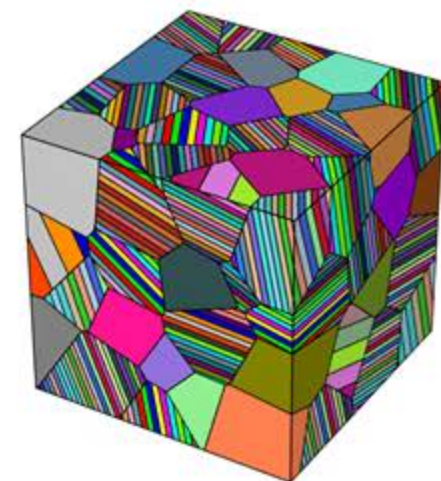


Globular grains

Voronoi based polycrystal model



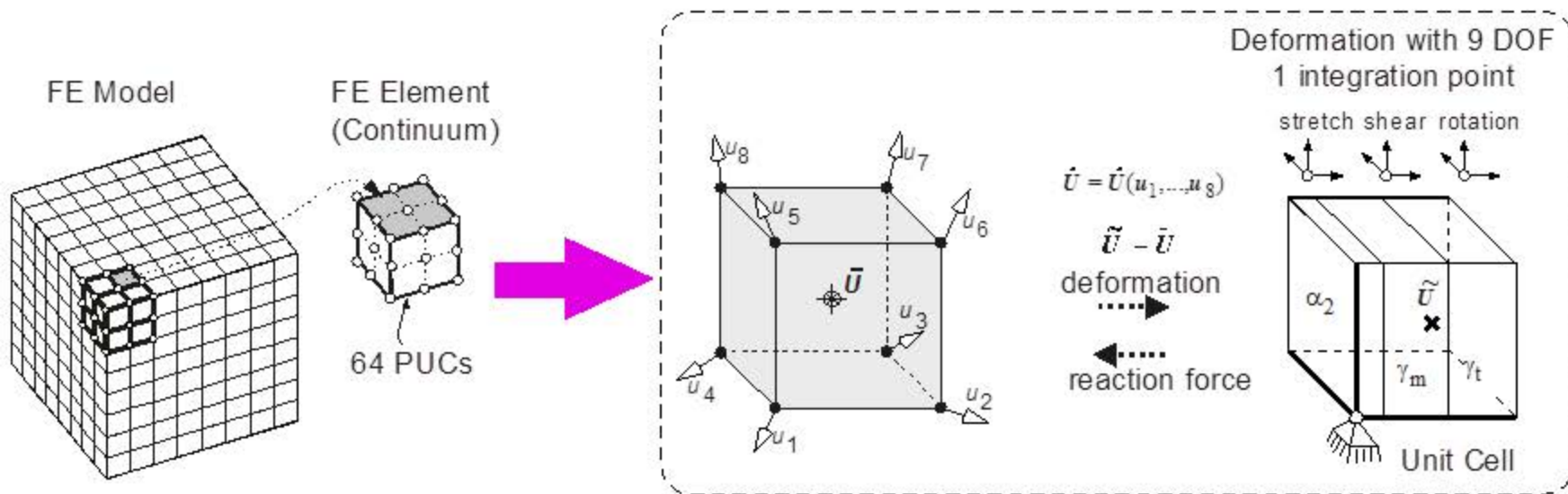
Lamellar microstructure



Globular microstructure



# Global-local coupling for average mechanical behaviour



Werwer et. al. *Comp. Mat. Sci.* 19, 2000

Model validation and  
parameter estimation

- PST lamellar alloy
- Fully lamellar alloy
- Duplex alloy

References:

- Lebensohn et al., *Acta mat* 46, 1998
- Werwer et al., *Int J Plast* 22, 2006
- Kabir et al. *Acta Mat* 58, 2010
- Cornec, Kabir, *Mat Sci Engg A620*, 2014

Constitutive behaviour of  
TiAl phases  
Crystal plasticity model

Classical  
(Huang et al)

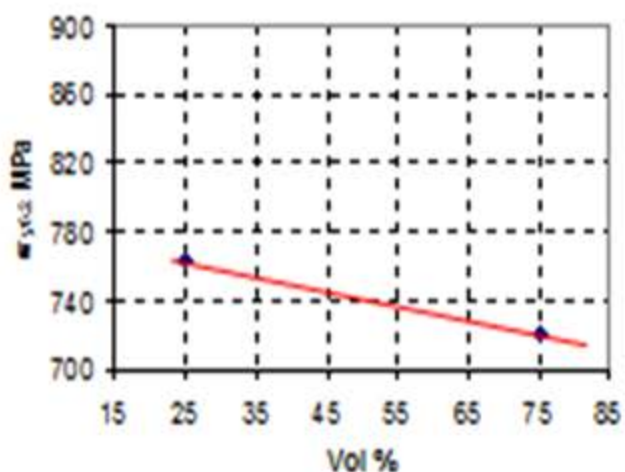
Gradient enhanced  
(Kabir, Shahid)

Temperature sensitive  
(Kabir, Ilyas)

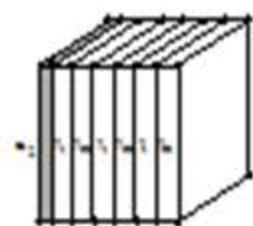
# Prediction: Microstructural influence on mechanical properties

## Microstructural influence on Yield

Influence of lamellar colonies vol%

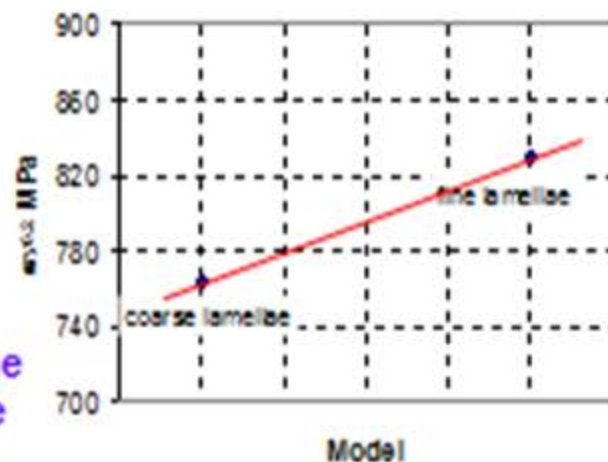


Change of Vol %

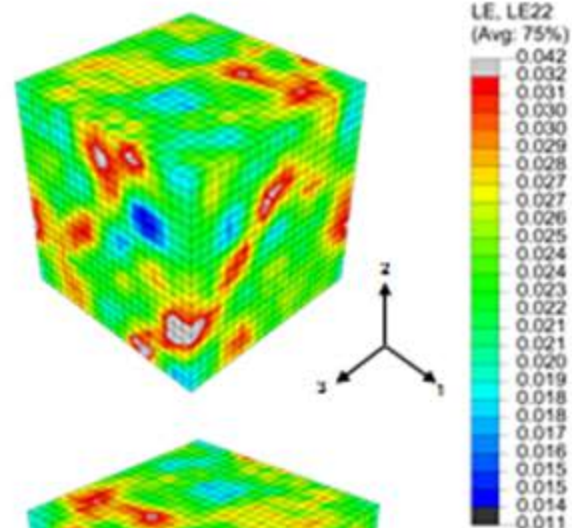


Unit cell with fine microstructure

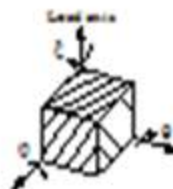
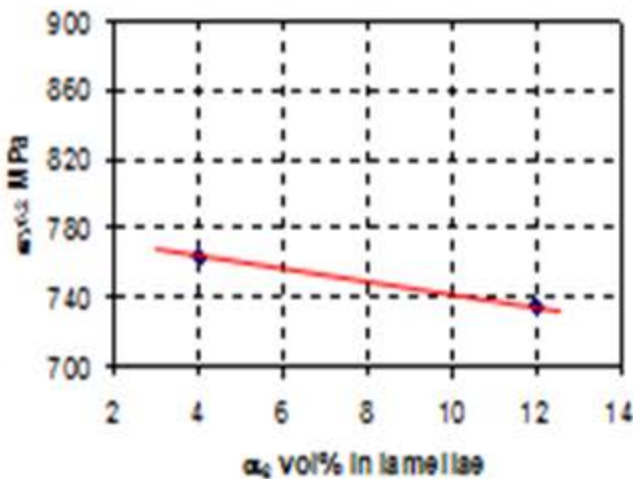
Duplex with coarse and fine lamellae



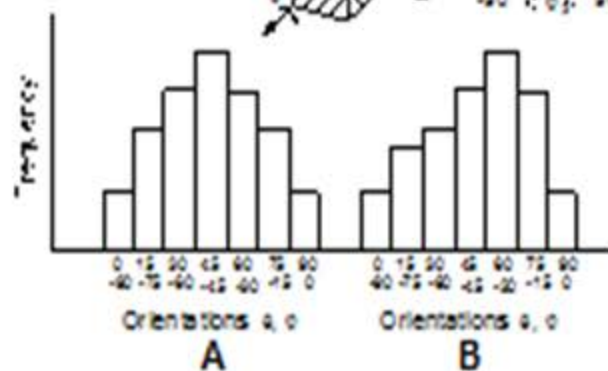
## Localized strain for lamellar orientation



Influence of phase volume %

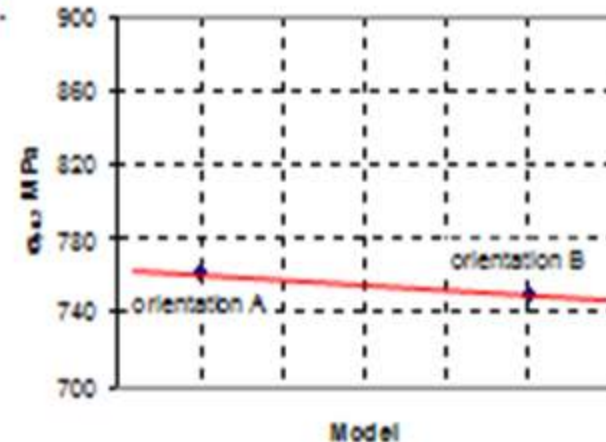


$$\begin{aligned} -90^\circ < \phi_1 < 90^\circ \\ 0^\circ < \phi_2 < 280^\circ \\ -90^\circ < \phi_3 < 90^\circ \end{aligned}$$



Orientation distribution of grains/lamellae

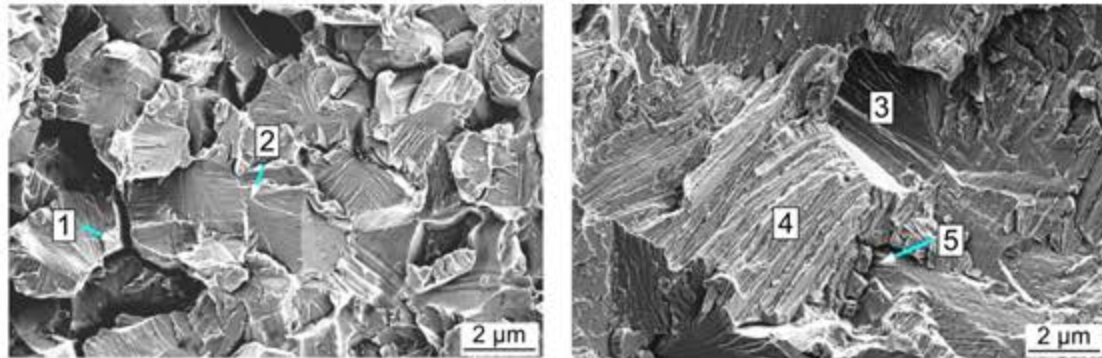
Orientation distribution: A and B





# Micromechanical modelling of crack initiation and propagation

Kabir et al., Mat Sci Eng A 635(2015)13-22



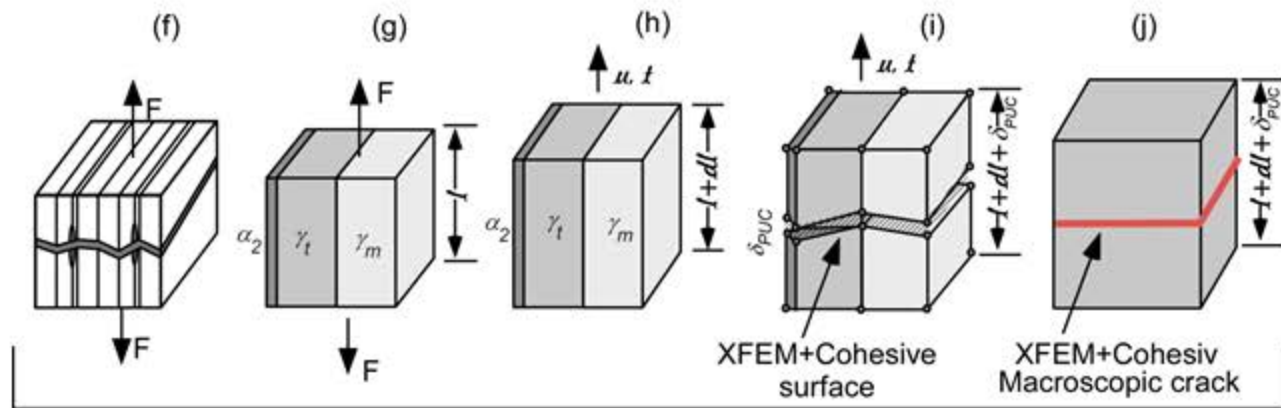
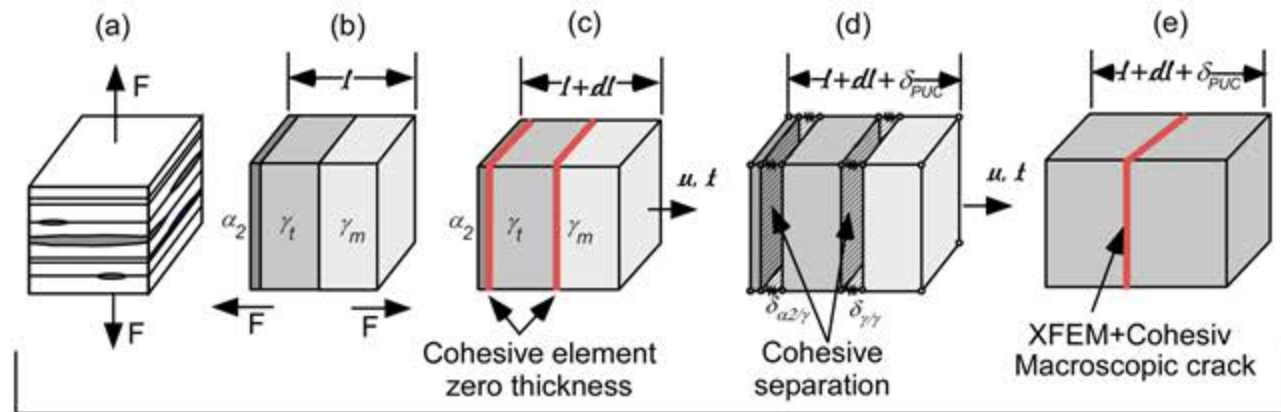
Q1 (annealed at 1230°C)

- 1 Inter-granular cracks
- 2 Trans-granular fracture

Q7 (annealed at 1290°C)

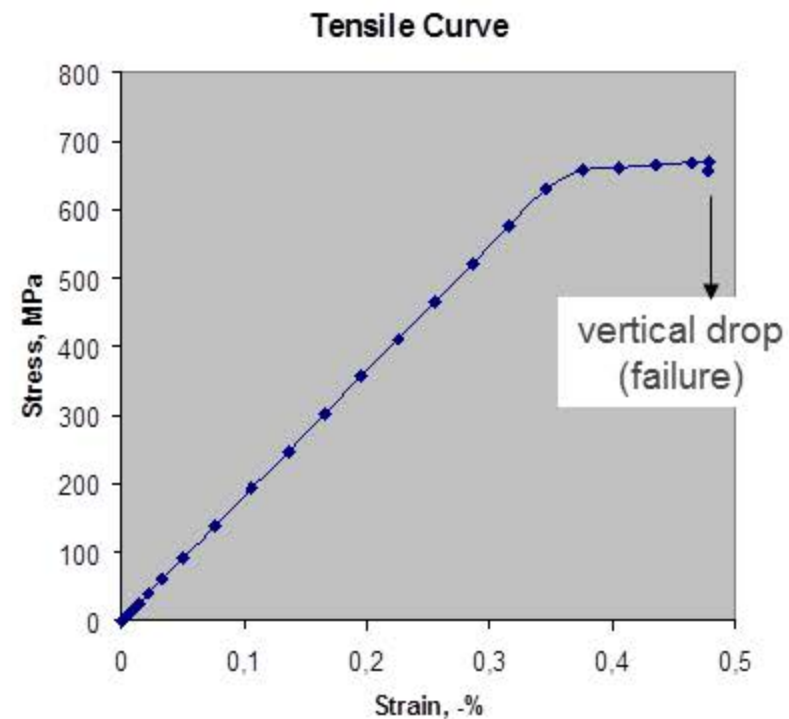
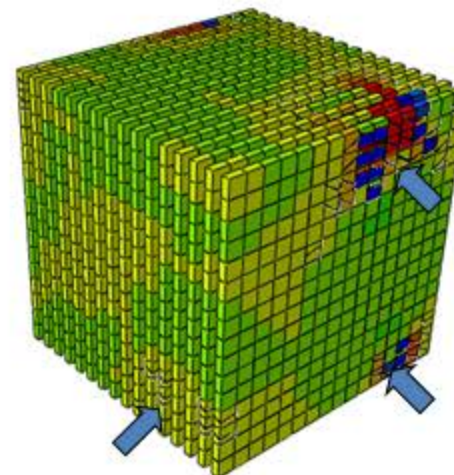
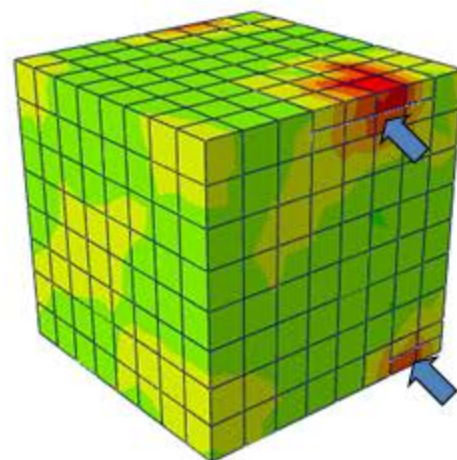
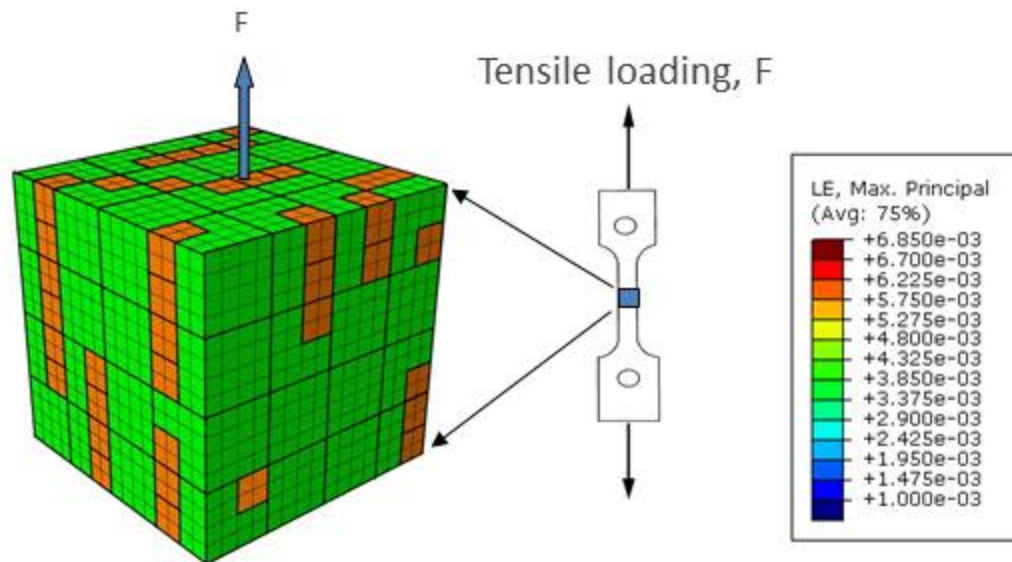
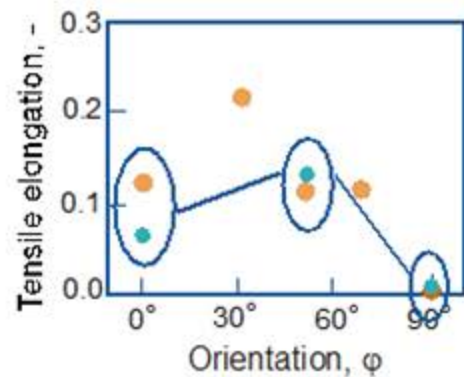
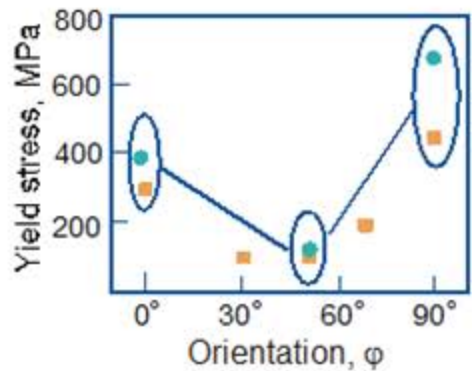
- 3 Trans-granular fracture
- 4 Trans-lamellar fracture
- 5 Cracking at colony boundary

- ❑ Two basic mechanisms:
- ❑ Interlamellar fracture, grain boundary fracture
  - ❑ Interface failure/debonding
- ❑ Translamellar fracture, Transgranular fracture
  - ❑ Splitting of material



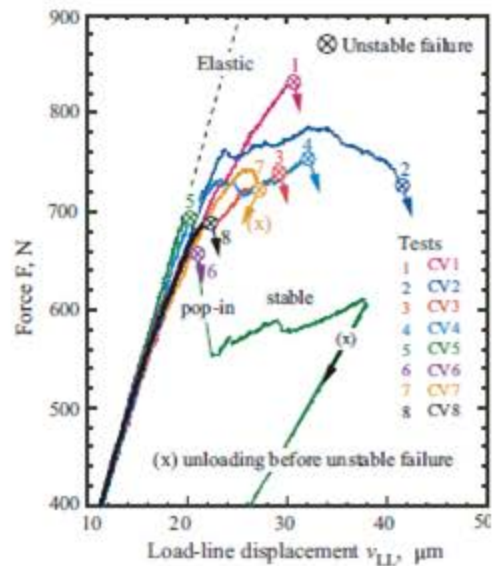
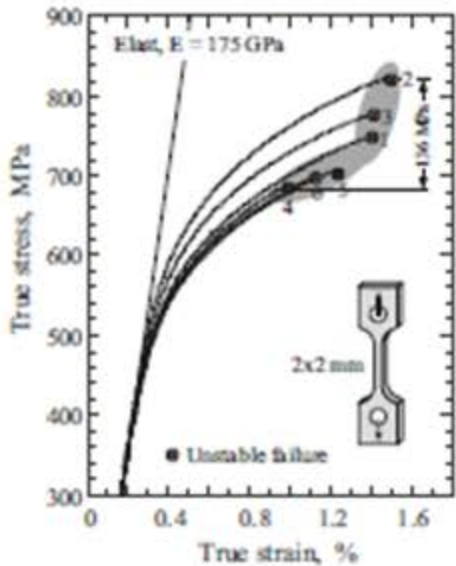
# Micromechanical modelling of crack initiation and propagation

Single crystal PST alloy: **Ti-49.8 Al (at.%)**  
 K.-F. YAO, H. INUI, K. KISHIDA and M. YAMAGUCHI  
*Acta metall. mater.* 43 (1995) pp. 1075

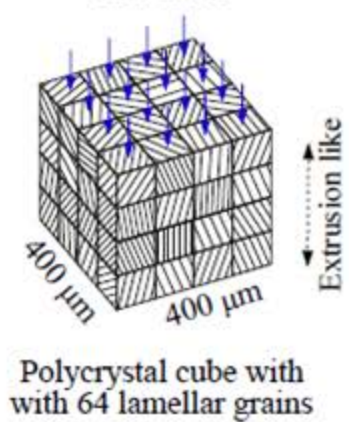




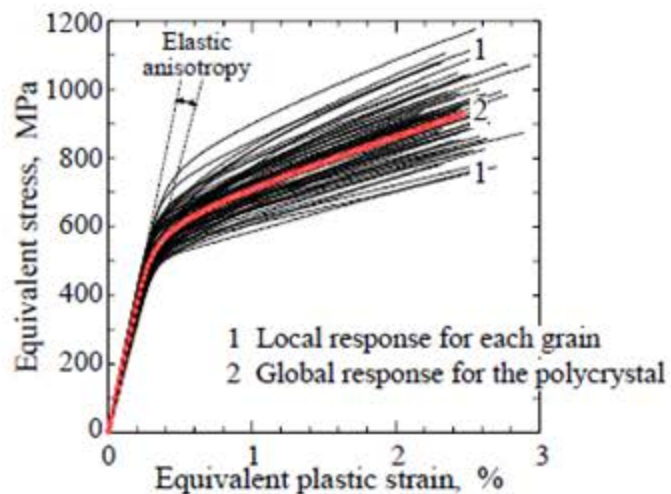
# Prediction of component failure using stochastic data



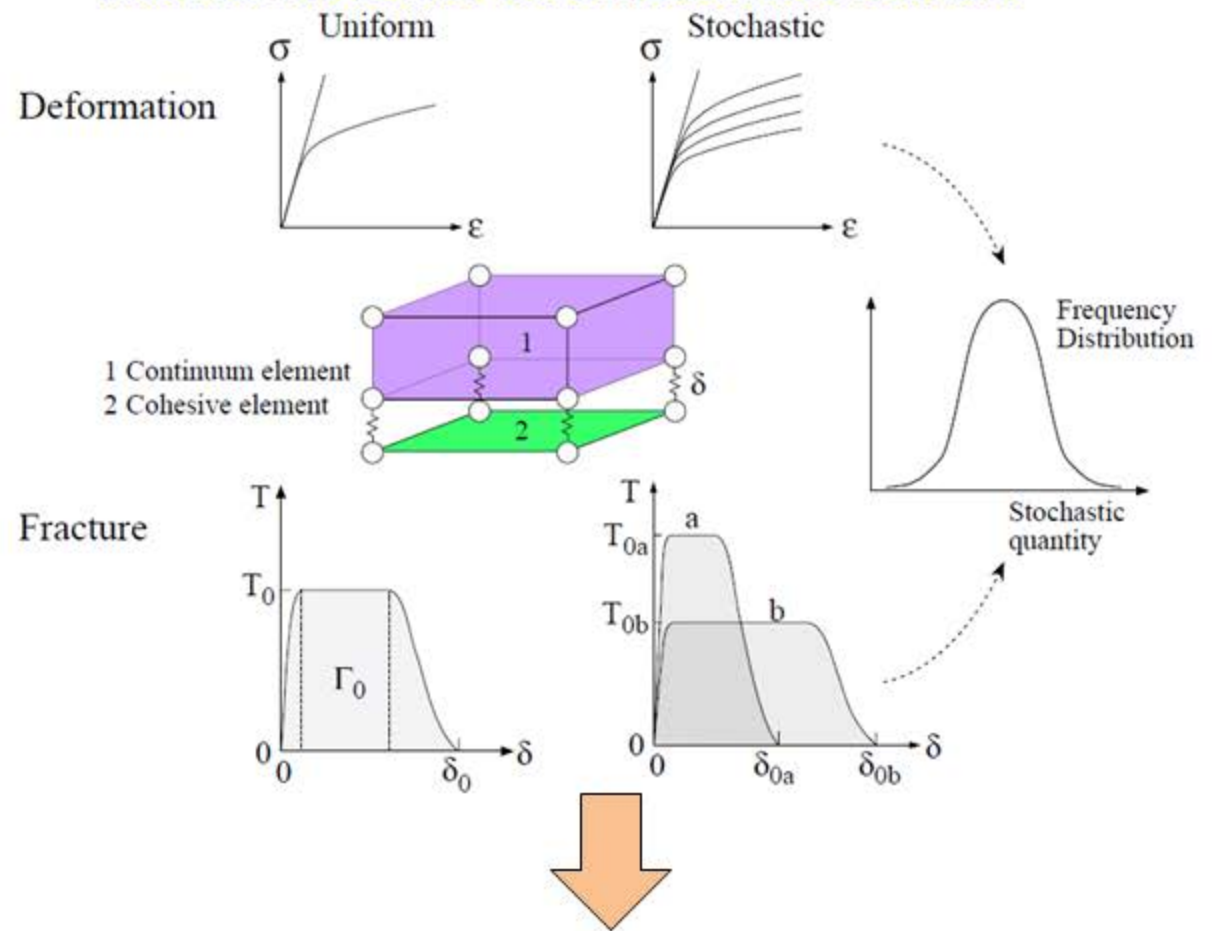
## Polycrystal FE Cube



## Simulation Response



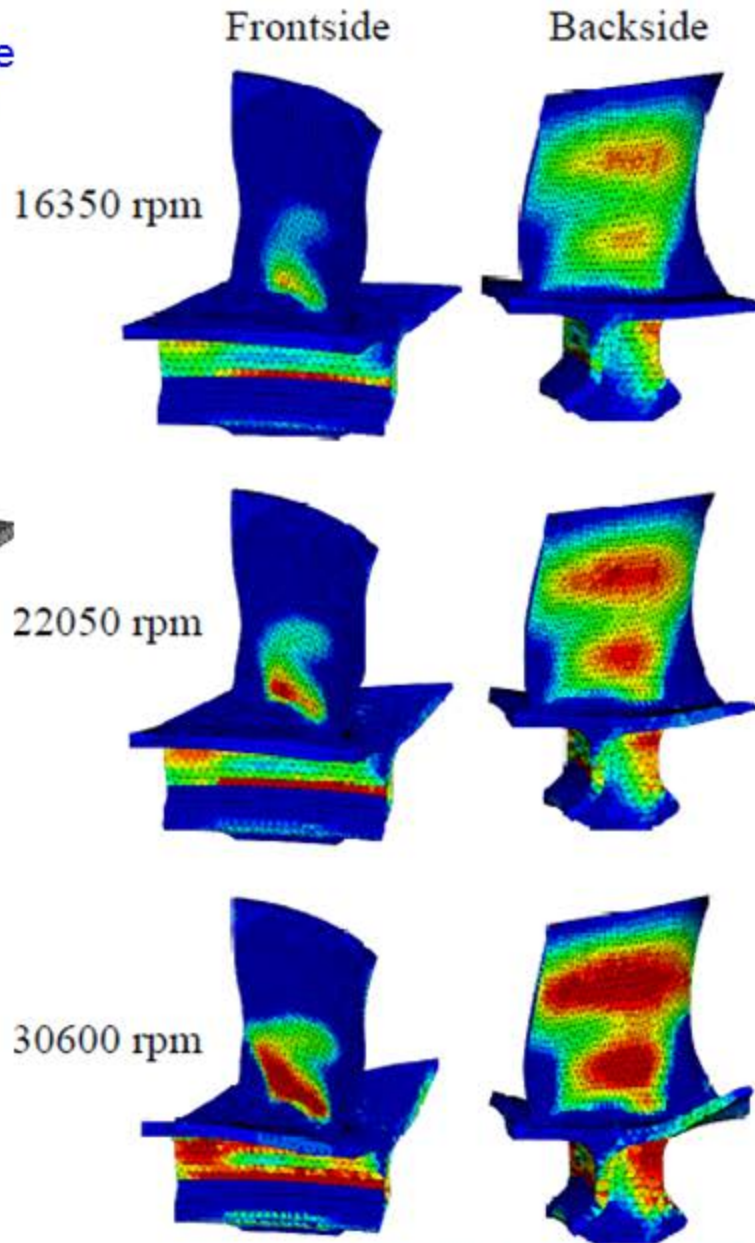
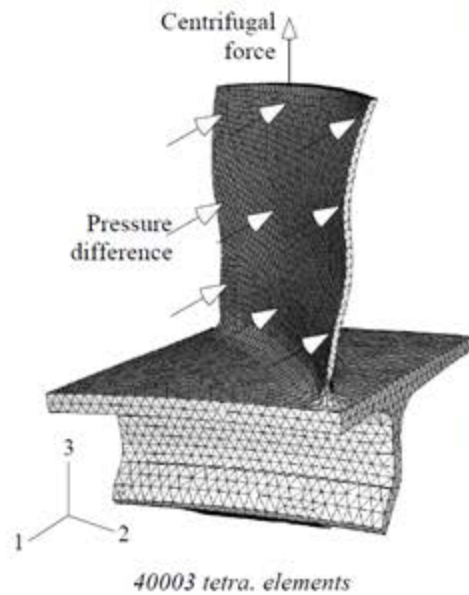
## Material Model for deformation and fracture



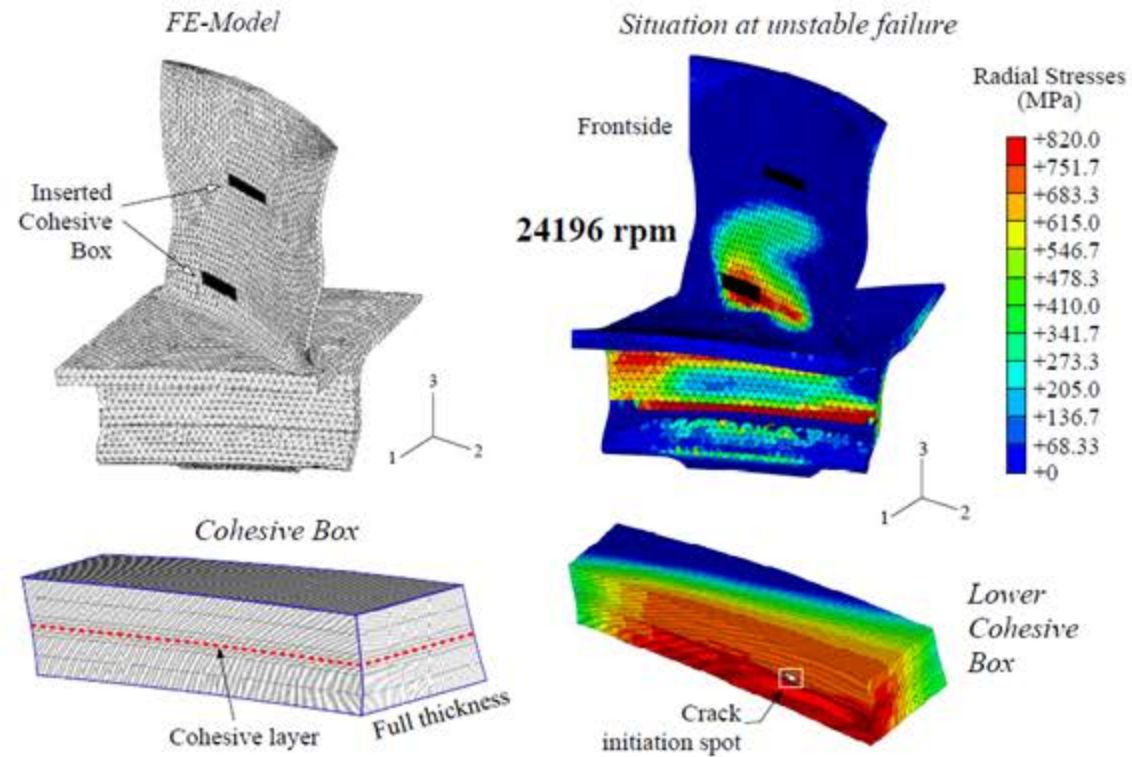
Analysis of maximum allowable RPM for compressor blade

# Component failure: Demonstration on a model-compressor blade

Test case with simple boundary condition



Analysis of maximum allowable RPM for compressor blade



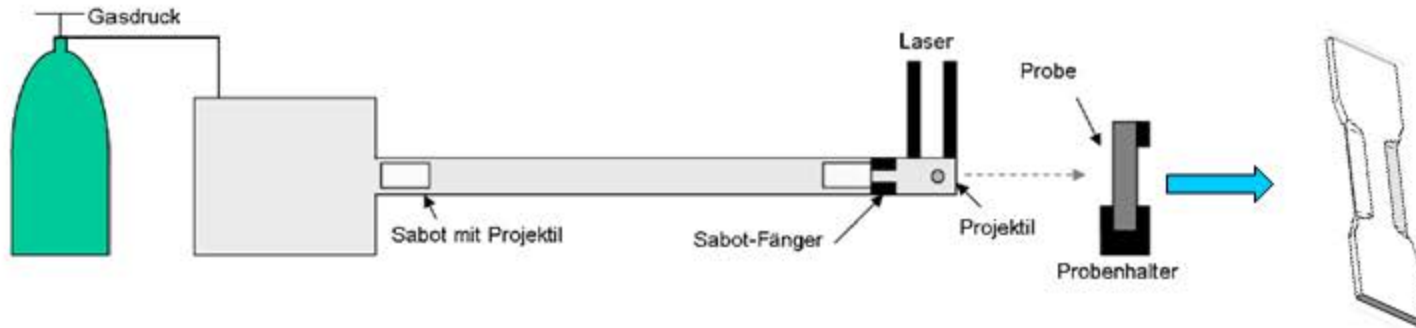


# Dynamic fracture: High velocity particle Impact on TiAl alloy

- References: 1) Susanne Gebhard, PhD Thesis, University of Stuttgart, 2011  
2) Gebhard et al, Materials Science and Engineering A 527 (2010) 5883–5891

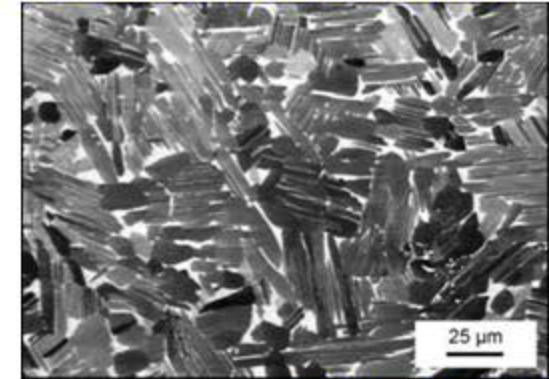
Goal:  
To understand domestic object damage (DOD) on turbine blade

High velocity particle Impact experiment  
Institute of Structures and Design, DLR, Stuttgart



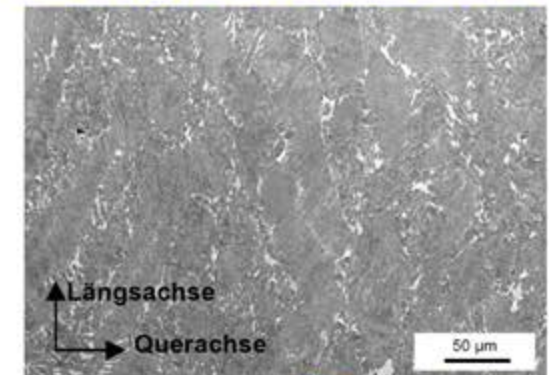
Tapered edge specimen  
(turbine blade edge)

As Cast microstructure



Globular grains  
 $\beta$ -phases at boundaries

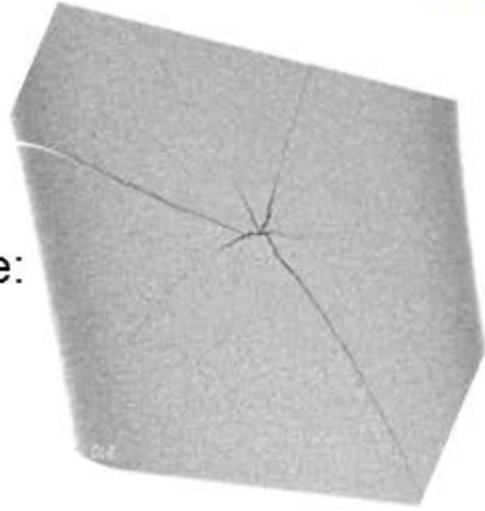
As forged microstructure



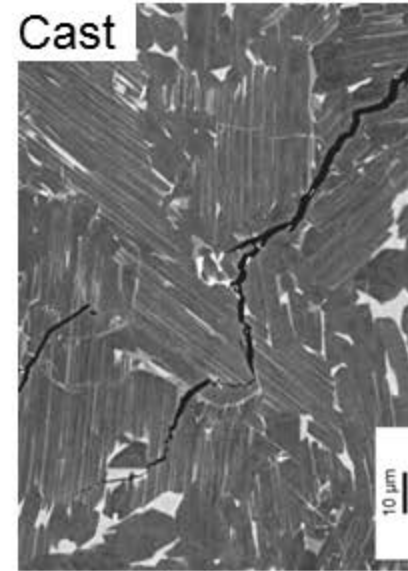
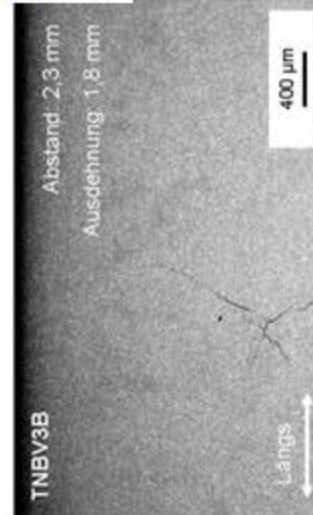
Elongated grains  
 $\beta$ -phases at boundaries

# High velocity particle Impact: Microstructure influence on backside crack-network

## As Cast microstructure

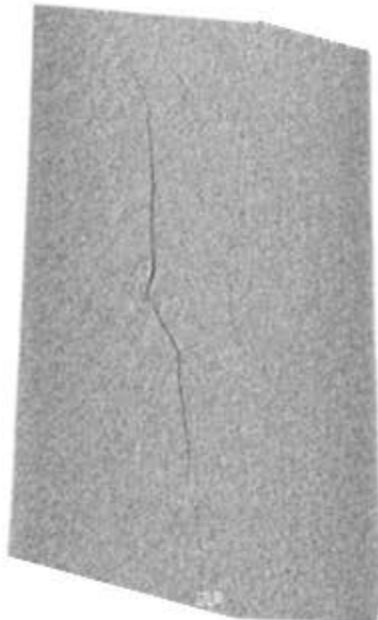


Crack profile:  
Star shape

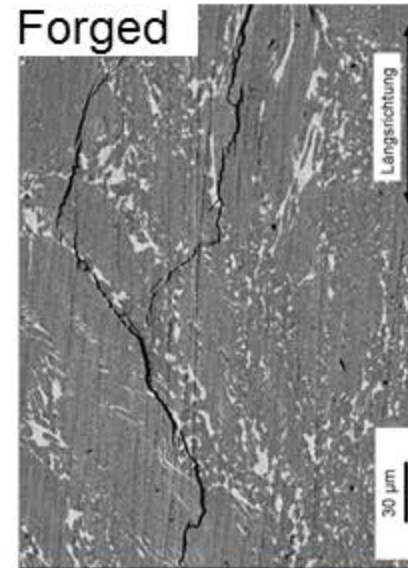


Fracture along grain boundaries in presence of  $\beta$ -phases

## As forged microstructure



Crack profile:  
elongated



- ❑ Analysis of residual strength
- ❑ Calculation of fatigue life

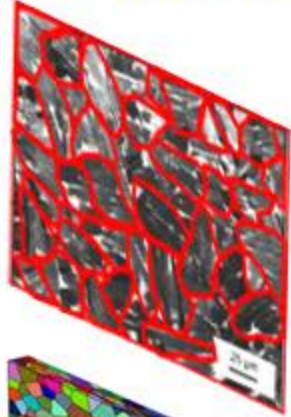
Crack length vs microstructure

Ref 1) S Gebhard, PhD Thesis, 2011

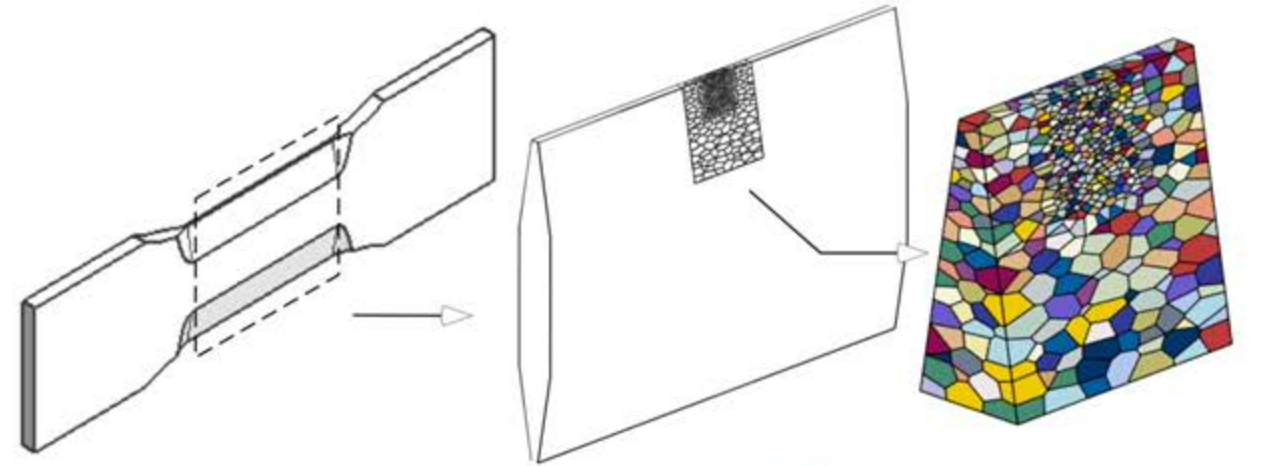
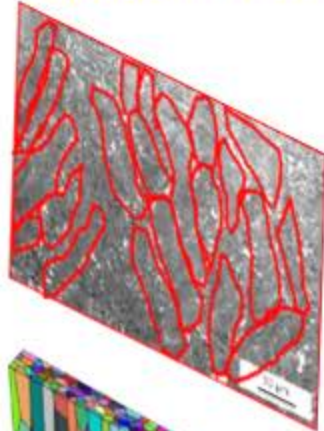


# Modelling of impact fracture: Synthetic (virtual) microstructures and FE model

As Cast  
microstructure



As forged  
microstructure



Tapered edge specimen  
(idealized turbine blade  
component)



Explicit grain and  
grain boundary  
modelling



Graded  
microstructure with  
graded mesh

- ❑ Voronoi based synthetic microstructure
  - ❑ Grain statistics are incorporated
- ❑ Microstructure simplification
  - ❑ Grains with ideal grain boundary
  - ❑ Grain boundary contains  $\beta$ -phases

Model generated with Neper[1]

[1] R. Quey, et al., Comp Methods in App Mech Eng, 2011 (200)1729



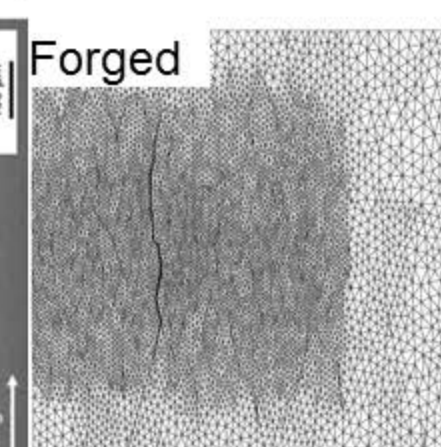
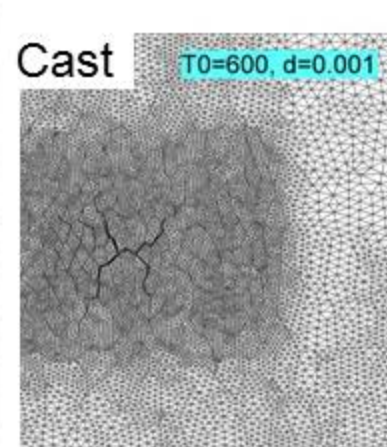
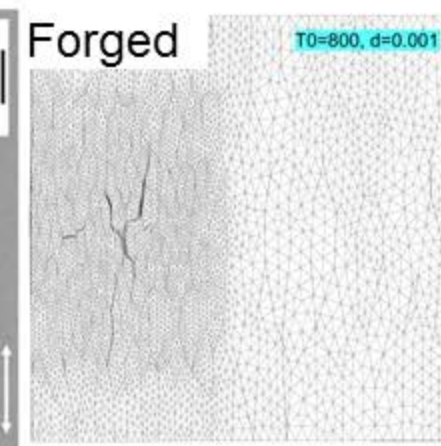
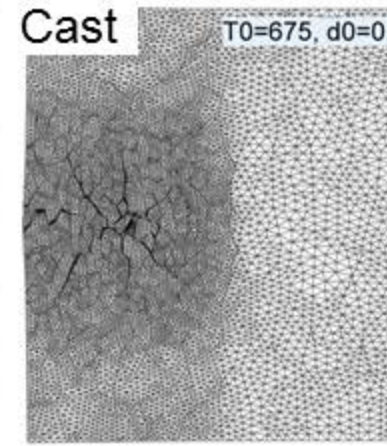
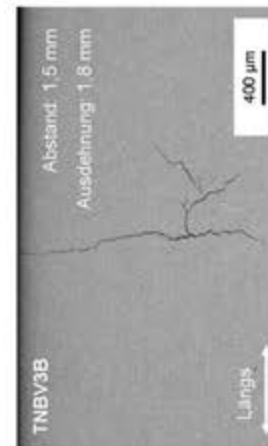
# Modelling of impact fracture: prediction of crack profile

Constitutive behaviour:  
Combined damage and fracture approach

For grain damage (tensile damage model)  
pressure stress > Hydrostatic cut-off stress  
(Deviatoric component remains zero)

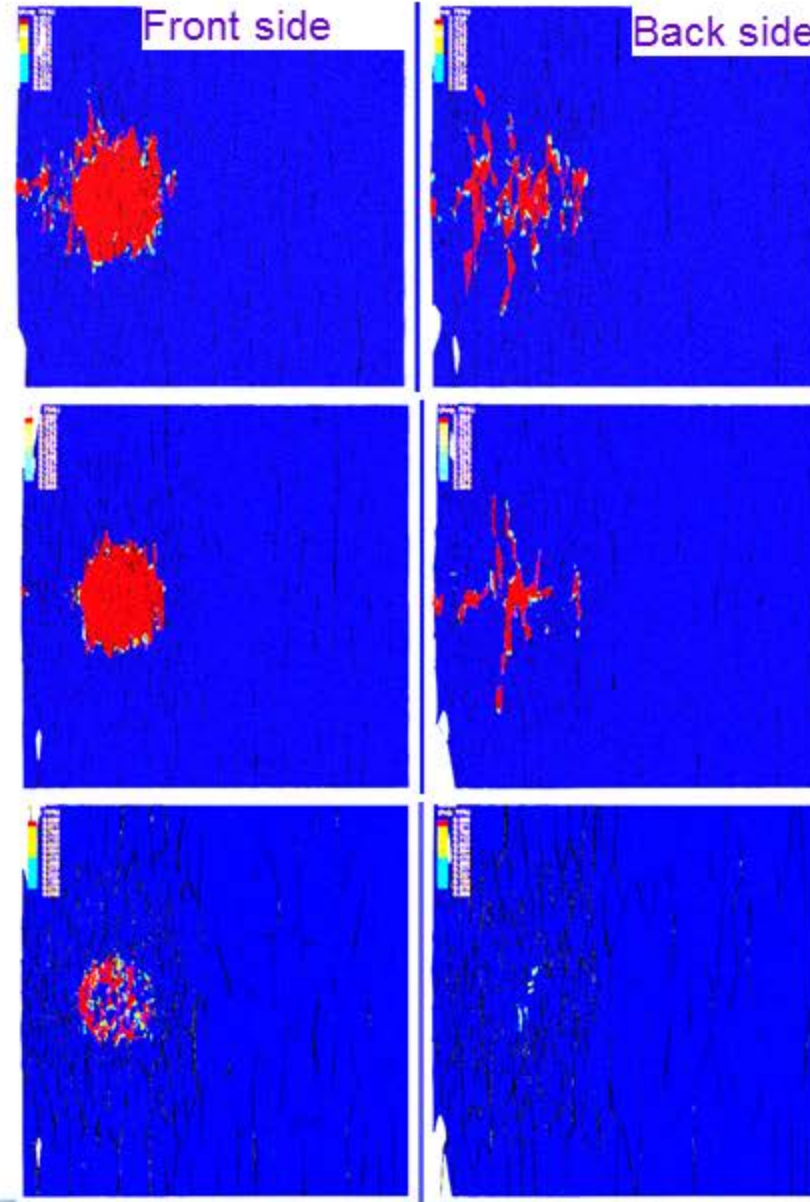
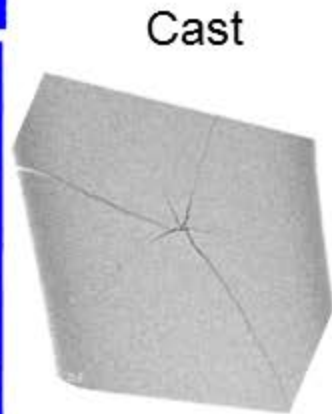
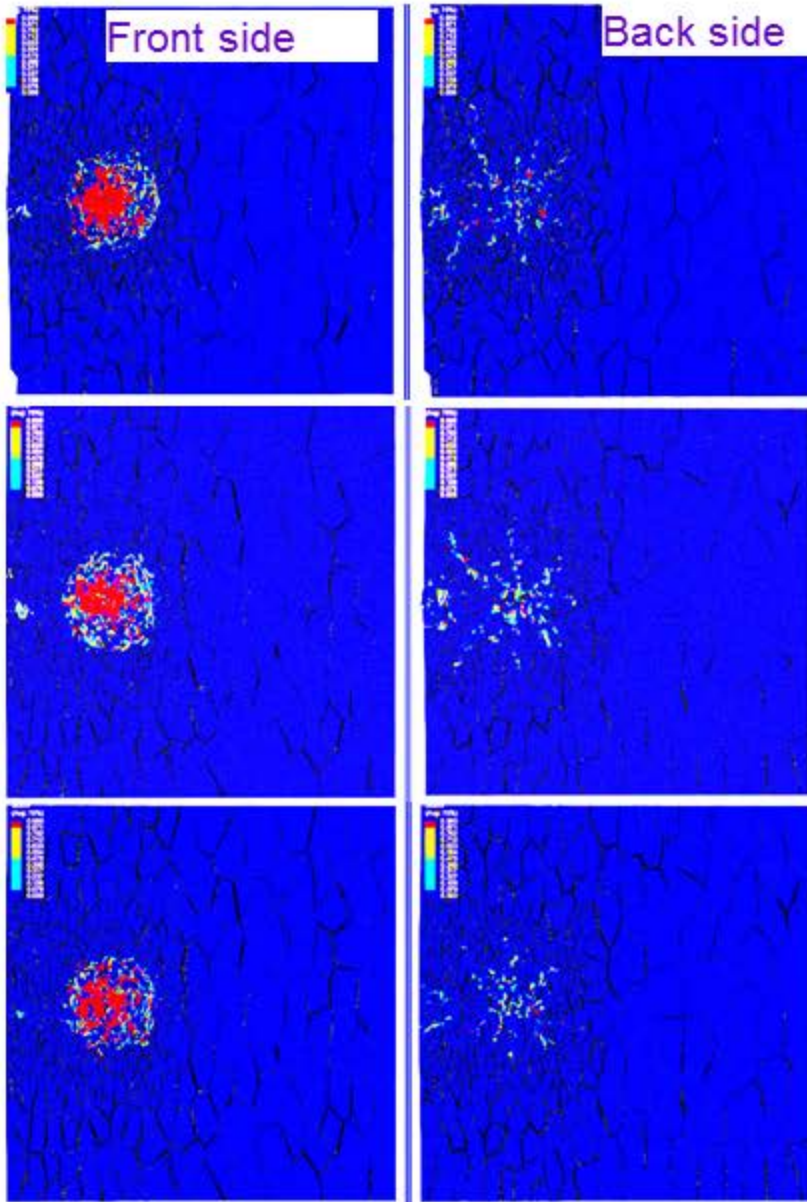
For interface damage (cohesive model):  
Interface strength > critical strength at interface  
Crack initiates at critical material separation

Microstructural sensitivity on back side crack network (Exp. and Sim.)



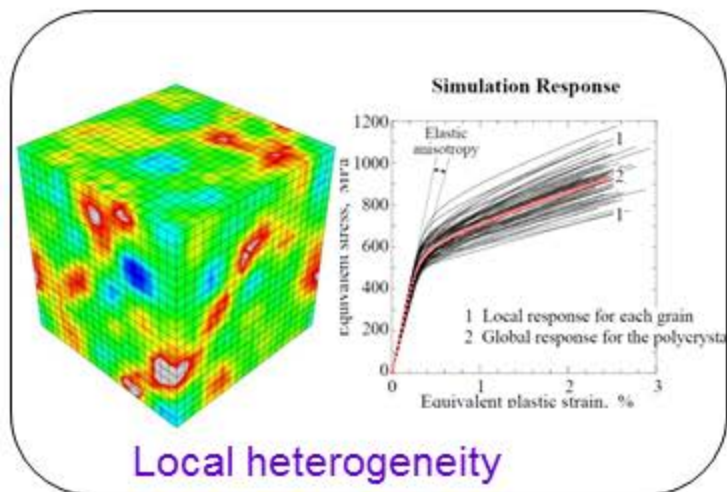


# Parameter study: crack profile for different damage parameters

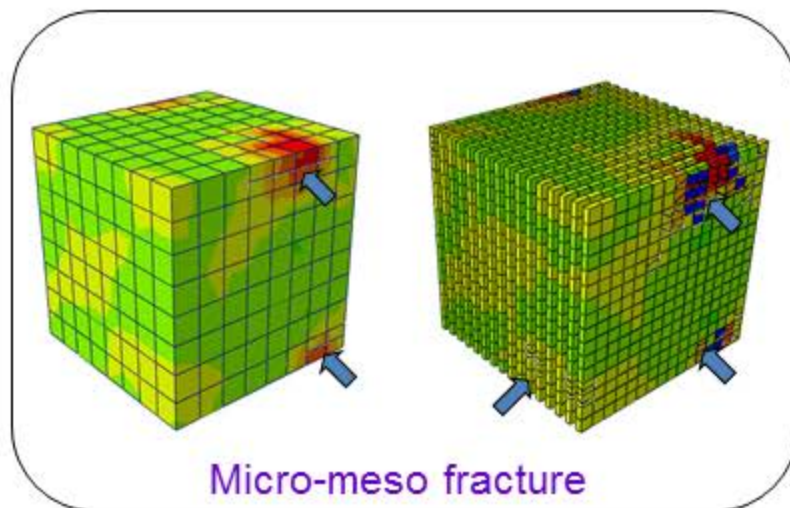


# Summary: Microstructure-sensitive modelling of TiAl alloy

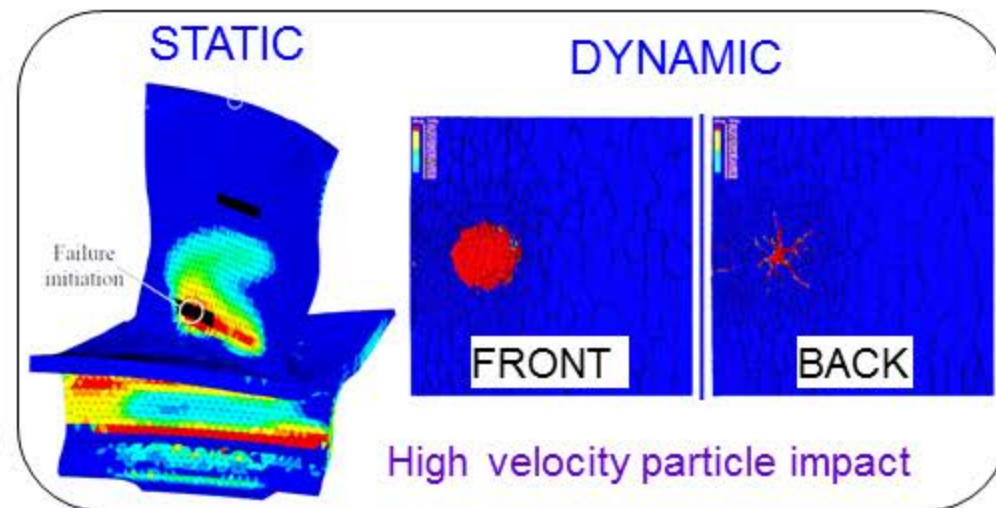
## Deformation



## Crack Propagation



## Fracture





# Thank you for your attention

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